

# Wrinkling of Orthotropic Viscoelastic Membranes

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This paper presents a simplified simulation technique for orthotropic viscoelastic membranes. Wrinkling is detected by a combined stress-strain criterion and an iterative scheme searches for the wrinkle angle using a pseudo-elastic material stiffness matrix based on a non-linear viscoelastic constitutive model. This simplified model has been implemented in ABAQUS/Explicit and is able to compute the behavior of a membrane structure by superposition of a small number of response increments. The model has been tested against a published solution for a time-independent isotropic membrane under simple shear and also against experimental results on StratoFilm 420 under simple shear.

## Nomenclature

$a_T, a_\sigma$	temperature, stress shift factors
$\widehat{C}_{ij}$	2D in-plane material stiffness matrix, Pa
$C_{ij}$	stiffness matrix rotated by angle $\alpha$ , Pa
$D$	creep compliance, 1/Pa
$D_0$	instantaneous compliance, 1/Pa
$D^{p,i}(t)$	pseudo-elastic compliance between time $i\Delta t$ and designated time $t$ , 1/Pa
$\bar{e}_i$	unit vector
$E$	relaxation modulus, Pa

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$E_0$	Young's modulus of membrane, N/mm <sup>2</sup>
$\widehat{E}^{p,i}$	in-plane pseudo-elastic stiffness matrix in directions of orthotropy, Pa
$E_w^{p,i}$	effective elasticity matrix for wrinkled element, Pa
$E_s^{p,i}$	effective elasticity matrix for slack element, Pa
$f$	friction, N
$F$	total shear force applied on membrane, N
$g_0, g_1, g_2$	nonlinear parameters for Schapery's viscoelastic model
$H$	height of membrane, mm
$L$	length of membrane, mm
$P$	total film tension, N
$S$	applied shear force, N
$S_{ij}$	2D compliance coefficient matrix
$t$	time, s; thickness of membrane, mm
$T$	temperature, K
$T_0$	reference temperature, K
$T_{ij}$	transformation matrix
$\alpha$	wrinkling angle, i.e. angle between direction of uniaxial stress and $x$ -axis, deg
$\delta_x$	shear displacement of membrane, mm
$\gamma$	shear strain of membrane
$\Delta D$	total transient compliance, 1/Pa
$\Delta D_i$	$i$ -th transient compliance at infinite time, 1/Pa
$\Delta D_{ij}$	2D in-plane compliance matrix, 1/Pa
$\epsilon$	Green strain tensor for fictitious wrinkled state
$\epsilon^u$	Green strain tensor state of natural uniaxial tension
$\tau_i$	$i$ -th relaxation time, s
$\psi$	reduced time, s
$\rho$	density of membrane, kg/mm <sup>3</sup>
$\nu$	Poisson's ratio of membrane
$\sigma_{\text{eff}}$	effective stress, Pa

## I. Introduction

Wrinkling of thin membranes has attracted much interest and yet little is known about wrinkles in anisotropic viscoelastic membranes. Our interest in this topic is motivated by current research in superpressure balloons, and particularly their behavior during inflation when the balloon envelope is heavily wrinkled. In recent papers<sup>1,2</sup> we have presented numerical simulations of some particular balloons that remain clefted when they are fully inflated and pressurized. These results were in agreement with experimental tests that had been conducted by the NASA Balloon Program, but the shape of the clefts in our simulations did not agree with the experimental observations. It was conjectured that these discrepancies may have resulted from the elastic and isotropic model for the balloon film that had been used. Hence it was decided to implement a more detailed material model that would allow for both anisotropy and visco-elasticity. However, when wrinkling had been combined with these effects we were unable to find any published test cases that could be used to confirm the validity of our simulations and so we decided to carry out our own set of experiments as well. We intend to study how these effects affect the shape of clefts in superpressure balloons, these results will be presented in future publications.

Here we present the outcome of our studies on wrinkling in moderately anisotropic, viscoelastic thin films. The paper begins with a literature review focused on a particular nonlinear viscoelastic material model for balloon film and an approach to the numerical simulation of wrinkling in anisotropic films. With this key background, in Section III we present a computational scheme that models the visco-elasticity of the film with an incremental, pseudo-elastic representation that is modified to allow for the formation of wrinkles, when a combined stress-strain wrinkling criterion is satisfied. This scheme is then implemented, in Section IV, as a VUMAT subroutine in the finite element software Abaqus/Explicit. Section V considers the special case of isotropic and elastic membranes, for which there is an extensive literature, and compares the results from our simulations to some published results. Section VI presents a set of experiments that have been carried out on balloon film in a simple shear apparatus; the experimental setup and procedure are described and the results of shear tests are presented. In Section VII these experimental results are compared with the simulations. Section VIII concludes the paper.

## II. Background

This section provides a compact review of viscoelasticity and wrinkling theories for orthotropic thin polymeric films.

## A. Review of Viscoelasticity Theory

In the classical theory of linear viscoelasticity<sup>3,4</sup> the creep compliance  $D$  and relaxation modulus  $E$  can be obtained from spring and dashpot models. A spring in series with a dashpot is known as the Maxwell model; a spring in parallel with a dashpot is known as the Kelvin-Voigt model. Both models have serious shortcomings. A more realistic model is the Standard Linear Solid model which is either represented by a Maxwell element with an additional spring in parallel or a Kelvin-Voigt element with an additional spring in series.

A more realistic representation of actual viscoelastic behavior is obtained by considering  $n$  Kelvin-Voigt elements with one spring in series, which leads to a compliance function expressed as a sum of exponential terms called Prony series

$$D(t) = D_0 + \sum_{i=1}^n \Delta D_i (1 - e^{-t/\tau_i}) \quad (1)$$

Viscoelastic materials are temperature as well as time dependent but both effects can be combined through the time-temperature superposition principle by considering only the reference temperature  $T_0$  and the reduced time

$$\psi(t) \equiv \int_0^t \frac{d\tau}{a_T(T)} \quad (2)$$

where  $a_T(T)$  is the temperature shift factor, defined to be 1 at the reference temperature ( $a_T(T_0) = 1$ ).

The compliance master curve, which defines the time-dependent compliance at the reference temperature, can be obtained from a series of uniaxial creep tests. It can be expressed as a combination of an instantaneous part  $D_0$  and a transient part  $\Delta D(\psi)$ :

$$D(\psi) = D_0 + \Delta D(\psi) \quad (3)$$

Here  $D_0$  represents the instantaneous compliance at the reduced time  $\psi = 0$  and the transient part  $\Delta D(\psi)$  can be represented by the Prony series:

$$\Delta D(\psi) = \sum_{i=1}^n \Delta D_i (1 - e^{-\psi/\tau_i}) \quad (4)$$

The linear viscoelastic strain at time  $t$ ,  $\epsilon(t)$ , is calculated by the convolution integral (also called Boltzman Superposition integral)

$$\epsilon(t) = D_0 \sigma(t) + \int_{0+}^t \Delta D(\psi^t - \psi^\tau) \frac{d\sigma(\tau)}{d\tau} d\tau \equiv D(\psi) \sigma(t) \quad (5)$$

Experimental studies have shown that this linear model is acceptable at small strains; at larger strains a nonlinear viscoelastic model has to be adopted, where the creep compliance is a function of stress, as well as of time and temperature. Following Schapery<sup>5,6</sup> this behavior can be captured by introducing an experimentally derived stress-related shift  $a_\sigma$  in the reduced time expression, which kicks in at stresses higher than a limiting value  $\sigma_0$ . Hence Equation 2 for the reduced time is rewritten as

$$\psi(t) \equiv \int_0^t \frac{d\tau}{a_T(T)a_\sigma(\sigma, T)} \quad (6)$$

Schapery<sup>5</sup> has adopted a single-integral representation for the strain at time  $t$  based on three nonlinear parameters  $g_0$ ,  $g_1$  and  $g_2$ :

$$\epsilon(t) = g_0 D_0 \sigma(t) + g_1 \int_{0+}^t \Delta D(\psi^t - \psi^\tau) \frac{dg_2 \sigma(\tau)}{d\tau} d\tau \equiv D(\psi) * \sigma(t) \quad (7)$$

where the nonlinear parameters are all equal to 1 in the linear viscoelasticity range.

The above uniaxial model has been extended<sup>5</sup> to plane stress loading of a polymeric film by making the assumption that the transient compliance in any direction can be expressed in terms of a constant matrix  $S_{ij}$  multiplied by the compliance in a direction of orthotropy of the material, assumed to be known

$$\Delta D_{ij} = S_{ij} \Delta D_{11} \quad (8)$$

where  $S_{11} = 1$  by definition. The uniaxial stress  $\sigma$  used to calculate all stress-dependent non-linearities in the previous model is then replaced with an effective stress. This is defined as

$$\sigma_{\text{eff}} = \sqrt{\widehat{\sigma_{11}}^2 + 2A_{12}\widehat{\sigma_{11}}\widehat{\sigma_{22}} + A_{22}\widehat{\sigma_{22}}^2 + A_{66}\widehat{\tau_{12}}^2} \quad (9)$$

where  $\widehat{\sigma_{11}}$  is the normal stress in the one of the directions of material orthotropy, and it is assumed that the uniaxial model has been developed in this particular direction,  $\widehat{\sigma_{22}}$  is the normal stress in the other direction of orthotropy, and  $\widehat{\tau_{12}}$  is the shear stress. The notation  $\widehat{\phantom{x}}$  denotes components in the directions of orthotropy.

## B. Wrinkling of Orthotropic Films

There is an extensive literature on wrinkling of thin elastic films. Key contributions were made by Stein and Hedgepeth,<sup>7</sup> Miller and Hedgepeth<sup>8</sup> and Mansfield;<sup>9</sup> a comprehensive review can be found in Wong and Pellegrino.<sup>10–12</sup> Here we shall focus specifically on wrinkling in orthotropic **viscoelastic** films, for which the literature is rather scarce.

Relevant previous work includes an analysis of wrinkling of viscoelastic membranes by

Jenkins and Leonard<sup>13</sup> using a relaxed energy and dissipation function in a finite element analysis scheme. On the analytical front, a generalized tension field theory for anisotropic membranes with restricted types of non-linear stress-strain relationships was proposed by Mansfield.<sup>14</sup> There is also a large amount of literature, starting from Reissner,<sup>15</sup> in which *artificial orthotropy* was introduced as a numerical trick to derive compression-free solutions for isotropic membranes.

Wu<sup>16–18</sup> represented the true deformed surface of a wrinkled membrane by a smooth pseudo-deformed surface and hence introduced the wrinkle strain in this surface as a kinematic variable that measures the wrinkliness of the actual wrinkled surface. Based on the same basic ideas, Kang and Im<sup>19</sup> proposed an iterative scheme convenient for the finite element analysis of wrinkling in orthotropic membrane structures, which was later adopted by Gerngross and Pellegrino.<sup>20–22</sup> Alternative approaches have been proposed by Epstein and Forcinito<sup>23</sup> and Raible et al.<sup>24</sup>

The general idea in Kang and Im’s paper is that a wrinkled region in a thin film is in a state of uniaxial tension and the orientation and magnitude of this tension can be obtained from an invariant relation between strain components. The actual wrinkled shape of the film is not of interest, instead a fictitious non-wrinkled surface is used to describe the average deformation of the film.

Consider an undeformed material element ABCD, Figure 1(a). Let  $(\hat{x}, \hat{y})$  denote a Cartesian coordinate system in the undeformed configuration, where  $\hat{x}, \hat{y}$  are aligned with the directions of orthotropy of the material. Let  $(x, y)$  denote a rotated Cartesian coordinate system for the same material point but such that  $x$  is the direction of the uniaxial stress in the wrinkled material. The rigid body rotation angle,  $\alpha$ , from  $\hat{x}\hat{y}$  to  $xy$  is called the *wrinkling angle*. It is noteworthy that for isotropic materials the wrinkling angle is the same as the principal stress or strain angle but this is generally not the case for anisotropic materials.

After this rigid body rotation, the material element, which is now considered in a rotated Cartesian coordinate system, is transformed from the undeformed configuration ABCD to the final, wrinkled configuration A''B''C''D'' in two steps, Figure 1(b). The first step is a pure deformation from the undeformed configuration ABCD to the deformed configuration A'B'C'D', due to the application of the correct uniaxial stress  $\sigma_x$ . No wrinkling occurs during this deformation, because there are no constraints on the deformation kinematics. This deformation consists of a normal strain  $\epsilon_x^u$ , a transverse contraction  $\epsilon_y^u$  (due to the Poisson’s ratio of the membrane), and an additional shear strain  $\gamma_{xy}^u$  (only in the case of anisotropic materials). This strain state is called the *state of natural uniaxial tension* and the superscript  $u$  has been used to denote the corresponding strains. The second step involves pure wrinkling from A'B'C'D' to A''B''C''D''. During this step, the stress in the material does not change and hence its strain state also does not change, but the material element

“shrinks” due to the formation of a series of wrinkles. At this point, though, the actual material element (which is no longer planar) is replaced by a fictitious wrinkled surface with the same outer profile.

Let  $\bar{e}_1$  and  $\bar{e}_2$  be unit vectors denoting respectively the  $x$  and  $y$  direction. The material point  $\bar{X} = X_1\bar{e}_1 + X_2\bar{e}_2$  in the reference configuration ABCD corresponds to point  $\bar{x} = x_1\bar{e}_1 + x_2\bar{e}_2$  in the state of natural uniaxial tension (i.e., in configuration A'B'C'D') and to point  $\bar{x}^u = x_1^u\bar{e}_1 + x_2^u\bar{e}_2$  in the final deformed configuration A''B''C''D''. The mapping from ABCD to A'B'C'D' is written as  $x_1^u = a^u X_1 + c^u X_2$  and  $x_2^u = b^u X_2$ , and the mapping from ABCD to A''B''C''D'' is written as  $x_1 = a X_1 + c X_2$  and  $x_2 = b X_2$ .

During the wrinkling process, the points A', B', C', D' move vertically to A'', B'', C'', D'', therefore there is no deformation of  $\bar{e}_2$ , but there is still contraction of  $\bar{e}_1$ , i.e.

$$a = a^u, c = c^u \quad \text{and} \quad b \leq b^u \quad (10)$$

The Green strain tensors for the fictitious wrinkled state and the state of natural uniaxial tension are

$$\begin{aligned} \epsilon &= \frac{1}{2}(a^2 - 1)\bar{e}_1 \otimes \bar{e}_1 + \frac{1}{2}ac(\bar{e}_1 \otimes \bar{e}_2 + \bar{e}_2 \otimes \bar{e}_1) + \frac{1}{2}(b^2 + c^2 - 1)\bar{e}_2 \otimes \bar{e}_2 \\ &= \epsilon_x \bar{e}_1 \otimes \bar{e}_1 + \epsilon_{xy} \bar{e}_1 \otimes \bar{e}_2 + \epsilon_{yx} \bar{e}_2 \otimes \bar{e}_1 + \epsilon_y \bar{e}_2 \otimes \bar{e}_2 \end{aligned} \quad (11)$$

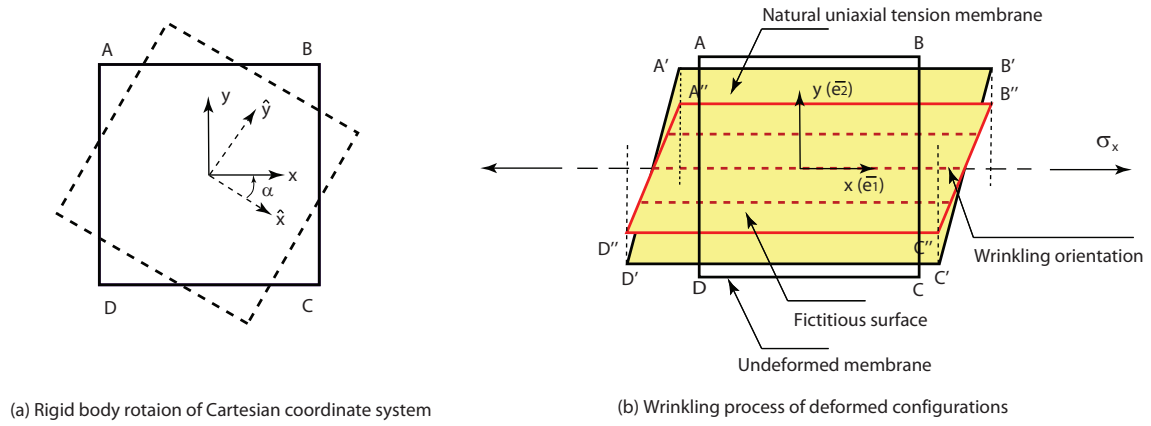
$$\begin{aligned} \epsilon^u &= \frac{1}{2}[(a^u)^2 - 1]\bar{e}_1 \otimes \bar{e}_1 + \frac{1}{2}(a^u)(c^u)(\bar{e}_1 \otimes \bar{e}_2 + \bar{e}_2 \otimes \bar{e}_1) + \frac{1}{2}[(b^u)^2 + (c^u)^2 - 1]\bar{e}_2 \otimes \bar{e}_2 \\ &= \epsilon_x^u \bar{e}_1 \otimes \bar{e}_1 + \epsilon_{xy}^u \bar{e}_1 \otimes \bar{e}_2 + \epsilon_{yx}^u \bar{e}_2 \otimes \bar{e}_1 + \epsilon_y^u \bar{e}_2 \otimes \bar{e}_2 \end{aligned} \quad (12)$$

Comparing the above three equations, we conclude that in the wrinkled state the strain components  $\epsilon_x$  and  $\gamma_{xy}$  remain unchanged, i.e.  $\epsilon_x = \epsilon_x^u$  and  $\gamma_{xy} = \gamma_{xy}^u$ , but the transverse strain  $\epsilon_y$  is different due to the over-contraction of the material in the direction perpendicular to the stress direction, i.e.  $\epsilon_y^u \geq \epsilon_y$ .

Assuming that the Green strain  $\epsilon$  in the membrane is small, it is approximately equal to the standard Cauchy strain and so the stress-strain relationship has the standard linear form

$$\begin{Bmatrix} \widehat{\sigma}_x \\ \widehat{\sigma}_y \\ \widehat{\tau}_{xy} \end{Bmatrix} = \begin{bmatrix} \widehat{C}_{11} & \widehat{C}_{12} & 0 \\ \widehat{C}_{21} & \widehat{C}_{22} & 0 \\ 0 & 0 & \widehat{C}_{66} \end{bmatrix} \cdot \begin{Bmatrix} \widehat{\epsilon}_x \\ \widehat{\epsilon}_y \\ \widehat{\gamma}_{xy} \end{Bmatrix} \quad (13)$$

which can be written in the compact form  $\{\widehat{\sigma}\} = [\widehat{C}]\{\widehat{\epsilon}\}$ . Note that the shear-extension coupling terms in this material stiffness matrix  $[\widehat{C}]$  are zero because the relationship has



**Figure 1. States of a wrinkled element.**

been set up in the directions of orthotropy of the material. This stress-strain relationship can be transformed to the  $x, y$  coordinate system by rotating through the angle  $\alpha$

$$\begin{Bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{Bmatrix} = \begin{bmatrix} C_{11} & C_{12} & C_{16} \\ C_{21} & C_{22} & C_{26} \\ C_{61} & C_{62} & C_{66} \end{bmatrix} \cdot \begin{Bmatrix} \epsilon_x \\ \epsilon_y \\ \gamma_{xy} \end{Bmatrix} \quad (14)$$

or  $\{\sigma\} = [C]\{\epsilon\}$ . Here  $\{\epsilon\}$  can be obtained from  $\{\hat{\epsilon}\}$  by the transformation

$$[C] = [T][\hat{C}][R][T]^{-1}[R]^{-1} \quad (15)$$

where  $[R] = \text{diag}\{1, 1, 2\}$  and

$$[T] = \begin{bmatrix} \cos^2(\alpha) & \sin^2(\alpha) & 2\sin(\alpha)\cos(\alpha) \\ \sin^2(\alpha) & \cos^2(\alpha) & -2\sin(\alpha)\cos(\alpha) \\ -\sin(\alpha)\cos(\alpha) & \sin(\alpha)\cos(\alpha) & \cos^2(\alpha) - \sin^2(\alpha) \end{bmatrix}. \quad (16)$$

The wrinkling angle  $\alpha$  has to be such that the normal stress in the direction perpendicular to the wrinkles is zero and the shear stress is also zero, that is  $\sigma_y^u = \tau_{xy}^u = 0$ . If this condition is satisfied, then the transverse strain  $\epsilon_y^u$  and the shear strain  $\gamma_{xy}^u$  can be obtained from

$$\epsilon_y^u = \epsilon_x^u \frac{C_{21}C_{66} - C_{26}C_{61}}{C_{26}C_{62} - C_{22}C_{66}} \quad (17)$$



$$\gamma_{xy}^u = \epsilon_x^u \frac{C_{22}C_{61} - C_{21}C_{62}}{C_{26}C_{62} - C_{22}C_{66}} \quad (18)$$

Kang and Im<sup>19</sup> proposed an iterative approach to search for the wrinkling angle; their procedure to find  $\alpha$  is as follows. First, one determines a value of  $\alpha$  that satisfies the relationship  $\gamma_{xy}^u = \gamma_{xy}$ , with  $\epsilon_x^u = \epsilon_x$ . Here,  $\epsilon_y^u$  and  $\gamma_{xy}^u$  are calculated from Equations 17 and 18. Finding  $\alpha$  is equivalent to solving the equation

$$f(\alpha) = \gamma_{xy}^u(\alpha) - \gamma_{xy}(\alpha) = 0 \quad (19)$$

which may have multiple solutions between 0 and 180 degrees. To find all of the solutions, we first divide the domain into ten uniform intervals and look for a change of sign between two boundary points in any interval. If a change of sign is found, then there is at least one solution inside this particular interval. Then we iteratively calculate the intersection between a straight line connecting two boundary points and the  $x$ -axis until the error is less than a prescribed tolerance. Once we have found a value of  $\alpha$  that satisfies this equation, we check that  $\epsilon_y^u \geq \epsilon_y$  and if this inequality is satisfied, then  $\alpha$  defines the wrinkling orientation. If no change of sign is found, the domain is divided into 50 intervals and the calculation is repeated.

### III. Effective Material Stiffness Matrix

A particular balloon film, StratoFilm 420, will be considered in this study. This film has been extensively characterized by Rand.<sup>25,26</sup> In the first part of this section we review the nonlinear viscoelastic properties of Stratofilm 420 and then develop an approximate pseudo-elastic constitutive model for a chosen temperature, time interval and stress level; this approach was inspired by Stubstad and Simitses.<sup>27</sup> In the second part of the section we modify the pseudo-elastic coefficients to consider the effects of wrinkling.

#### A. Pseudo-elastic model for StratoFilm 420

StratoFilm 420 is a 38 micron thick film made of Linear Low Density Polyethylene.<sup>28</sup> Rand<sup>25,26</sup> has developed a nonlinear viscoelastic constitutive model for this film.

The transient compliance  $\Delta D(T_0, \psi)$  in the machine direction of the film was given at a reference temperature  $T_0 = 293.16$  K and the instantaneous compliance  $D_0 = 3.0 \times 10^{-10}$  (Pa<sup>-1</sup>) was chosen to ensure that the transient compliance  $D_i$  would all be positive. The coefficients of a 15 term Prony series are listed in Table 2.

The temperature shift factor  $a_T$  was evaluated over the range 163 K to 323 K and then

**Table 2. Prony series for StratoFilm SF420**

$i$	$D_i$ (MPa $^{-1}$ )	$\tau_i$ (s)
1	$1.8764 \times 10^{-4}$	$1.6548 \times 10^{-16}$
2	$2.9249 \times 10^{-5}$	$4.8697 \times 10^{-15}$
3	$5.8224 \times 10^{-5}$	$1.4330 \times 10^{-13}$
4	$8.7542 \times 10^{-5}$	$4.2170 \times 10^{-12}$
5	$1.1561 \times 10^{-4}$	$1.2409 \times 10^{-10}$
6	$1.4159 \times 10^{-4}$	$3.6517 \times 10^{-9}$
7	$1.6989 \times 10^{-4}$	$1.0746 \times 10^{-7}$
8	$2.0924 \times 10^{-4}$	$3.1623 \times 10^{-6}$
9	$2.7274 \times 10^{-4}$	$9.3057 \times 10^{-5}$
10	$3.7796 \times 10^{-4}$	$2.7384 \times 10^{-3}$
11	$5.4670 \times 10^{-4}$	$8.0582 \times 10^{-2}$
12	$8.0581 \times 10^{-4}$	2.3714
13	$1.1844 \times 10^{-3}$	69.783
14	$1.7204 \times 10^{-3}$	2053.5
15	$2.6285 \times 10^{-3}$	60430

fitted at the reference temperature  $T_0 = 293.16\text{K}$  in the logarithmic form

$$\log a_T = \begin{cases} (T - 293.16)[7.33 \times 10^{-4}(T - 273.16) - 0.179133] & T > 233.16 \\ 3.1068 - 0.2350275(T - 273.16) & T < 233.16 \end{cases} \quad (20)$$

The stress shift was represented by  $a_\sigma$

$$\log a_\sigma = -0.126(\sigma_{\text{eff}} - \sigma_0) \quad (21)$$

where  $\sigma_0$  is threshold at which linear behavior ends and nonlinear behavior begins

$$\sigma_0 = 69.527 - 0.430944T + 6.7962 \times 10^{-4}T^2 \quad (22)$$

Two of the non-linear coefficients,  $g_0$  and  $g_1$  were set equal to 1. The parameter  $g_2$  was described by

$$g_2 = 1 + 0.1875(\sigma_{\text{eff}} - \sigma_0) \quad (23)$$

The coefficients for the biaxial model were, in addition to  $S_{11} = 1$ ,

$$S_{22} = 1.122 + 6.5895 \times 10^{-4}T - 6.609 \times 10^{-6}T^2 \quad (24)$$

and  $S_{12} = S_{21} = -0.58$ ,  $S_{66} = 4.45$ .

The coefficients for the effective stress,  $A_{ij}$ , were expressed as

$$A_{22} = 1.44, A_{12} = -0.4 \text{ and } A_{66} = 0.8 \quad (25)$$

A finite-element implementation of the above constitutive model for the analysis of a single balloon lobe has been developed by the Gerngross and Pellegrino<sup>20-22</sup> model. Here we present an alternative approach that, although less accurate, is more suitable for large scale simulations of balloon structures.

For a given time  $t$ , the integral representation for the strain  $\epsilon(t)$  in Equation 7 can be approximated by replacing the convolution integral with a series of  $m$  equal substeps  $\Delta t = \frac{t}{m}$  where each term is evaluated at the discrete time  $t_i = i\Delta t$

$$\begin{aligned} \epsilon(t) &\approx g_0 D_0 \sigma(t) + g_1 \sum_{i=1}^m \Delta D(\psi^t - \psi^{t_{i-1}}) \Delta[g_2(\sigma^{t_i})\sigma(t_i)] \\ &= g_0 D_0 \sigma(t) + g_1 \sum_{i=1}^m \Delta D(\psi^t - \psi^{t_{i-1}}) [g_2(\sigma^{t_i})\sigma(t_i) - g_2(\sigma^{t_{i-1}})\sigma(t_{i-1})] \end{aligned} \quad (26)$$

Substituting  $g_0 = g_1 = 1$  and assuming that  $g_2$  changes slowly over  $\Delta t$ , i.e.  $g_2(\sigma^{t_i}) \approx g_2(\sigma^{t_{i-1}})$ , then the above equation becomes

$$\begin{aligned} \epsilon(t) &\approx D_0 \sum_{i=1}^m \Delta\sigma(t_i) + \sum_{i=1}^m g_2(\sigma^{t_i}) \Delta D(\psi^t - \psi^{t_{i-1}}) \Delta\sigma(t_i) \\ &= \sum_{i=1}^m [D_0 + g_2(\sigma^{t_i}) \Delta D(\psi^t - \psi^{t_{i-1}})] \Delta\sigma(t_i) \end{aligned} \quad (27)$$

Hence, we can define a pseudo-elastic compliance,  $D^{p,i}$  (the superscript  $p$  corresponds to "pseudo-elastic") between  $i\Delta t$  and  $t$ , and considering the corresponding reduced times

$$D^{p,i} = D_0 + g_2 \Delta D(\psi^t - \psi^{t_{i-1}}) \quad (28)$$

The corresponding strain can then be expressed as

$$\epsilon(t) = \sum_{i=1}^m D^{p,i} \Delta\sigma(t_i) = \sum_{i=1}^m \Delta\epsilon(t_i) \quad (29)$$

This uniaxial expression can be generalized to plane stress by defining the pseudo-elastic compliance in the machine direction,  $\widehat{D}_{11}^{p,i}$ , from Equation 28

$$\widehat{D}_{11}^{p,i} = D_0 + g_2 \Delta D(\psi^t - \psi^{t_{i-1}}) \quad (30)$$

and the remaining coefficients of the pseudo-elastic compliance matrix are then

$$\widehat{D}_{22}^{p,i} = S_{22}\widehat{D}_{11}^{p,i}, \quad \widehat{D}_{12}^{p,i} = \widehat{D}_{21}^{p,i} = S_{12}\widehat{D}_{11}^{p,i}, \quad \widehat{D}_{66}^{p,i} = S_{66}\widehat{D}_{11}^{p,i} \quad (31)$$

It follows from Equation 29 that the strain at time  $t$  can be obtained as a sum of strain increments, which are calculated from the stress increments multiplied by the corresponding total pseudo-elastic compliances. For each increment, the pseudo-elastic compliance is constant.

For example, if we consider the conditions of constant temperature  $T = 293$  K, constant stress  $\widehat{\sigma}_x = 4.5$  MPa,  $\widehat{\sigma}_y = 5.0$  MPa and  $\widehat{\tau}_{xy} = 0$  MPa, the shift of the compliance master curve in the machine direction is shown in Figure 2. At time  $t = 1000$  s, the pseudo-elastic moduli can be obtained from the compliance at the point marked with a dot in Figure 2. They are respectively  $\widehat{E}_{11}^p = 90.98$  MPa and  $\widehat{E}_{22}^p = 121.68$  MPa.

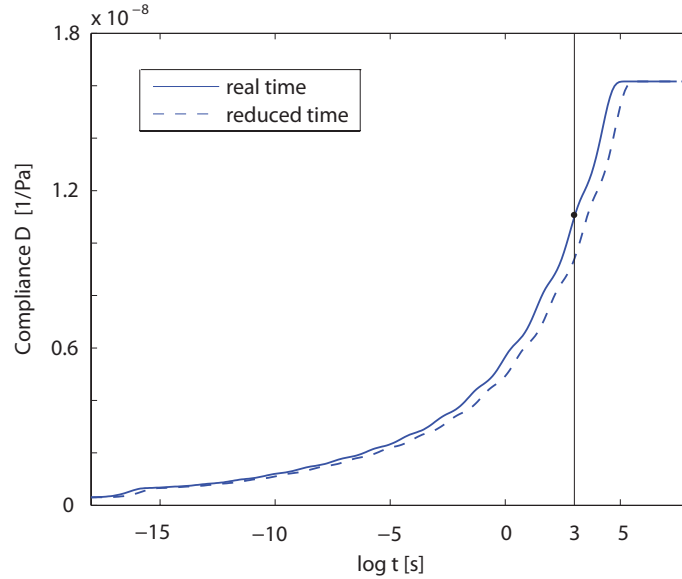


Figure 2. Master curves of compliance in machine direction.

The pseudo-elastic stiffness matrix in the directions of orthotropy of the film can be obtained by inversion of  $\widehat{D}^{p,i}$

$$\widehat{E}^{p,i} = \begin{bmatrix} \widehat{E}_{11}^{p,i} & \widehat{E}_{12}^{p,i} & 0 \\ \widehat{E}_{21}^{p,i} & \widehat{E}_{22}^{p,i} & 0 \\ 0 & 0 & \widehat{E}_{66}^{p,i} \end{bmatrix} \quad (32)$$

where

$$\begin{aligned}
\hat{E}_{11}^{p,i} &= \frac{1}{D_{11}^{p,i}} \frac{1}{S_{11} - S_{12}^2/S_{22}} \\
\hat{E}_{12}^{p,i} &= E_{21}^{p,i} = \frac{1}{D_{11}^{p,i}} \frac{1}{S_{12} - S_{11}S_{22}/S_{12}} \\
\hat{E}_{22}^{p,i} &= \frac{1}{D_{11}^{p,i}} \frac{1}{S_{22} - S_{12}^2/S_{11}} \\
\hat{E}_{66}^{p,i} &= \frac{1}{D_{11}^{p,i}} \frac{1}{S_{66}}
\end{aligned}$$

## B. Correction for Wrinkling

Kang and Im<sup>19</sup> compared three types of wrinkling criteria, the principal stress criterion, the principal strain criterion and the mixed criterion. They concluded that the mixed criterion works better for anisotropic wrinkling as it avoids numerical divergence caused by making wrong judgements about the membrane state, which happened with the principal stress based criterion.

The effective elasticity matrix depends on the state of a membrane element:

- Minor principal stress  $\sigma_2 > 0$ : the membrane is *taut*, hence Equation 32 is valid.
- Minor principal stress  $\sigma_2 \leq 0$  and major principal strain  $\varepsilon_1 > 0$ : the membrane is *wrinkled*.

Hence we must solve for the wrinkling angle  $\alpha$ . First we transform the pseudo-elastic stiffness in Equation 32 to the rotated coordinate system  $x, y$  (the transformation is analogous to Equation 14). In this system, the normal strain  $\epsilon_x$  and the shear strain  $\gamma_{xy}$  are equal to the strains in the state of natural uniaxial tension whereas  $\epsilon_y$  is not. The transverse stress  $\sigma_y$  has to be zero in a coordinate system aligned with the wrinkles, hence

$$\sigma_y = E_{21}^{p,i} \epsilon_x + E_{22}^{p,i} \epsilon_y + E_{23}^{p,i} \gamma_{xy} = 0 \quad (33)$$

From the above equation,  $\epsilon_y$  can be expressed in terms of  $\epsilon_x$  and  $\gamma_{xy}$ .

$$\epsilon_y = -\frac{E_{21}^{p,i}}{E_{22}^{p,i}} \epsilon_x - \frac{E_{26}^{p,i}}{E_{22}^{p,i}} \gamma_{xy} \quad (34)$$

Hence we eliminate  $\epsilon_y$  by substituting this expression into Equation 14, and obtain the

effective elasticity matrix<sup>22</sup> for the wrinkled element

$$E_w^{p,i} = \begin{bmatrix} E_{11}^{p,i} - \frac{E_{12}^{p,i} E_{21}^{p,i}}{E_{22}^{p,i}} & 0 & E_{13}^{p,i} - \frac{E_{12}^{p,i} E_{26}^{p,i}}{E_{22}^{p,i}} \\ 0 & 0 & 0 \\ E_{61}^{p,i} - \frac{E_{62}^{p,i} E_{21}^{p,i}}{E_{22}^{p,i}} & 0 & E_{66}^{p,i} - \frac{E_{26}^{p,i} E_{62}^{p,i}}{E_{22}^{p,i}} \end{bmatrix} \quad (35)$$

An alternative method is using directly the strains in the state of natural uniaxial tension  $\{\epsilon^u\}$  from Equations 17 and 18. Hence the corrected stress in the wrinkled state  $\{\sigma\}$  has to be a uniaxial tension, rewritten as  $\{\sigma\} = \{\sigma^u\} = E^{p,i}\{\epsilon^u\}$ . Since the element is under uniaxial tension, only the  $x$  component  $\sigma_x$  is nonzero. Finally  $\{\sigma\}$  is transformed back to  $\{\hat{\sigma}\}$  in the direction of orthotropy.

- Major principal strain  $\epsilon_1 \leq 0$  and minor principal stress  $\sigma_2 < 0$ : the membrane is *slack* and hence  $E_s^{p,i} = [0]$

In conclusion, to calculate the correct stress, the following steps are carried out:

- Step 1. Check the state of the membrane element by means of the mixed criterion;
- Step 2. If the element is wrinkled, find the wrinkling angle  $\alpha$  from Kang and Im's scheme, described in Section 2;
- Step 3. Compute the corrected Cauchy stress using the effective elasticity matrix.

The stress and strain components used for the above wrinkling tests should ideally be the best current estimates of the values at time  $t$ . However our current approach is to run each step  $i$  as a separate simulation in ABAQUS/Explicit and hence the only estimates that are available in each run are only the estimates based on the current increment.

## IV. Finite Element Implementation

In Section III we have presented an approach to obtain approximate estimates of the stresses in a viscoelastic film at time  $t$ . The loading sequence is divided into  $m$  steps, where  $m$  is determined by the required precision, and

$$\sigma(t) = \sum_{i=1}^m \Delta\sigma(t_i) = \sum_{i=1}^m \hat{E}^{p,i} \Delta\epsilon(t_i) \quad (36)$$

For example, the strain increments due to a three step loading sequence are shown on the left of Figure 3(a), hence

$$\epsilon = \Delta\epsilon_1 + \Delta\epsilon_2 + \Delta\epsilon_3 \quad (37)$$

The stress at the end time,  $t_3$ , due to the strain increments  $\Delta\epsilon_i$  is computed with ABAQUS/Explicit in three separate analyses with compliances  $\hat{E}^{p,i}$ . The results are shown in Figure 3(b-d) and are superposed in Figure 3(a). From the plots on the right in Figure 3, the stress at time  $t_3$  is given by

$$\begin{aligned}\sigma(t_3) &= \Delta\sigma_1 + \Delta\sigma_2 + \Delta\sigma_3 \\ &= E^{p,1}\Delta\epsilon_1 + E^{p,2}\Delta\epsilon_2 + E^{p,3}\Delta\epsilon_3\end{aligned}\tag{38}$$

Note that an approximation in this approach is to assume that the wrinkling directions in each step will not change significantly between the various steps,. Also, the coefficient  $g_2$  is stress-dependent and so is not correctly estimated for the full stress level. In other words, the proposed linear superposition is suitable for weakly nonlinear materials with only small wrinkle angle changes.

A user-defined material ABAQUS/Explicit subroutine (VUMAT) has been written to model wrinkling in an orthotropic viscoelastic film, it is schematically described in Figure 4.

This subroutine can be downloaded from

[http://pellegrino.caltech.edu/super\\_pressure\\_balloons.html](http://pellegrino.caltech.edu/super_pressure_balloons.html).

## V. Verification for Isotropic Elastic membranes

We have tested our VUMAT implementation of wrinkling in ABAQUS/Explicit by means of a comparison with the analytical solution for a linear-elastic, time-independent rectangular membrane under simple shear. The longer edges of the membrane are attached to rigid edges that are sheared uniformly while the shorter edges are unconstrained. This problem was investigated by Wong and Pellegrino.<sup>12</sup>

### A. Construction of FE Model

This test case involves a rectangular membrane under simple shear.<sup>29</sup> The properties of the film used in the model and the geometrical dimensions of the rectangular membrane are summarized in Table 3.

The membrane was modeled with 3-node, fully integrated triangular membrane elements (M3D3). All translations of the bottom edge nodes were fully constrained, whereas the top edge nodes were left free. As shown in Figure 5, the shear load was applied in terms of a horizontal shear displacement of the top edge.

The loading process consisted of two analysis steps: during the first step, lasting 2.5 seconds, the upper edge nodes were moved by 3 mm in the  $x$ -direction while all other translations were constrained. In the second step, also lasting 2.5 seconds, all the translational

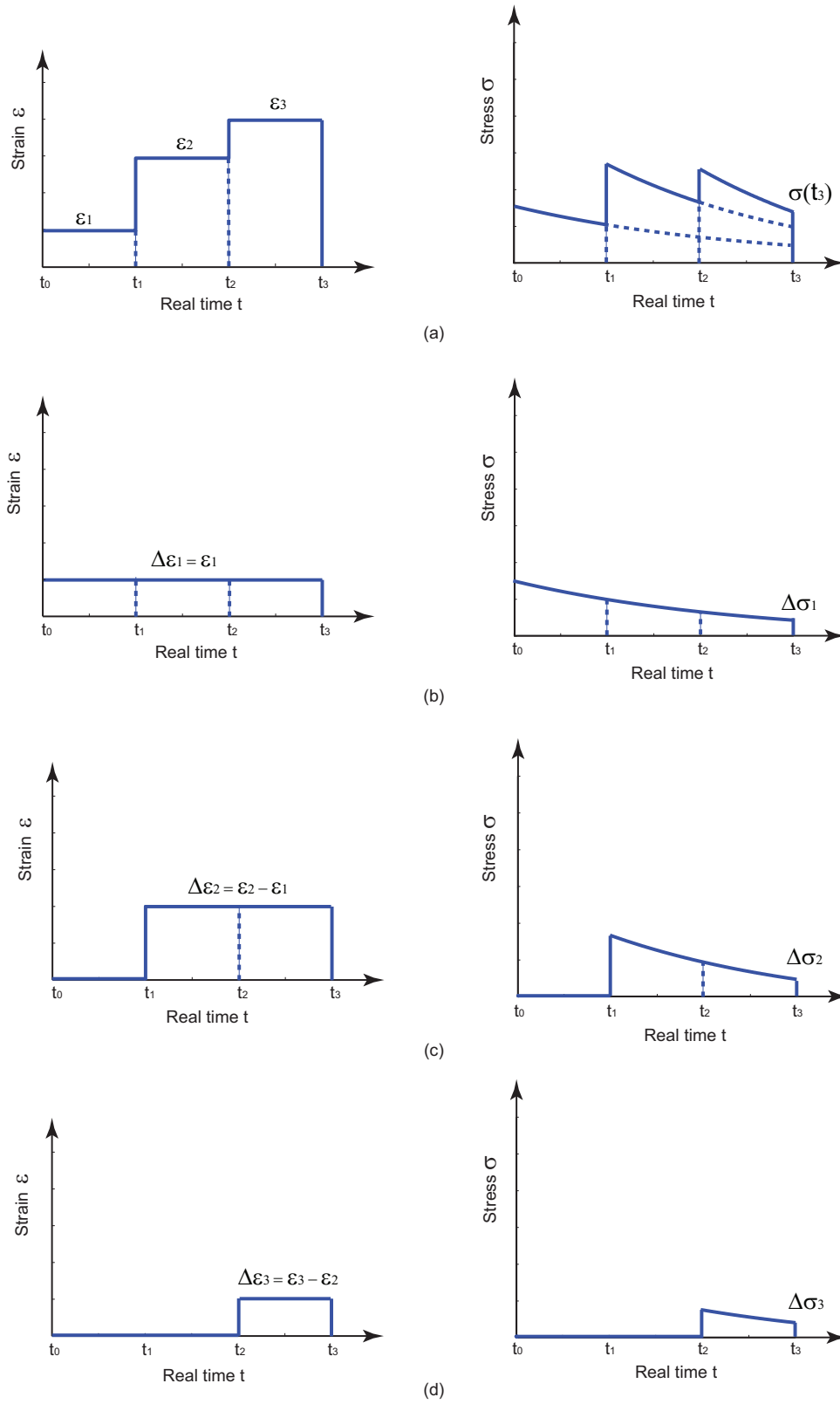


Figure 3. Superposition of loading steps: (a) total strain input and stress response; (b)–(d) incremental strain functions and corresponding stress responses.



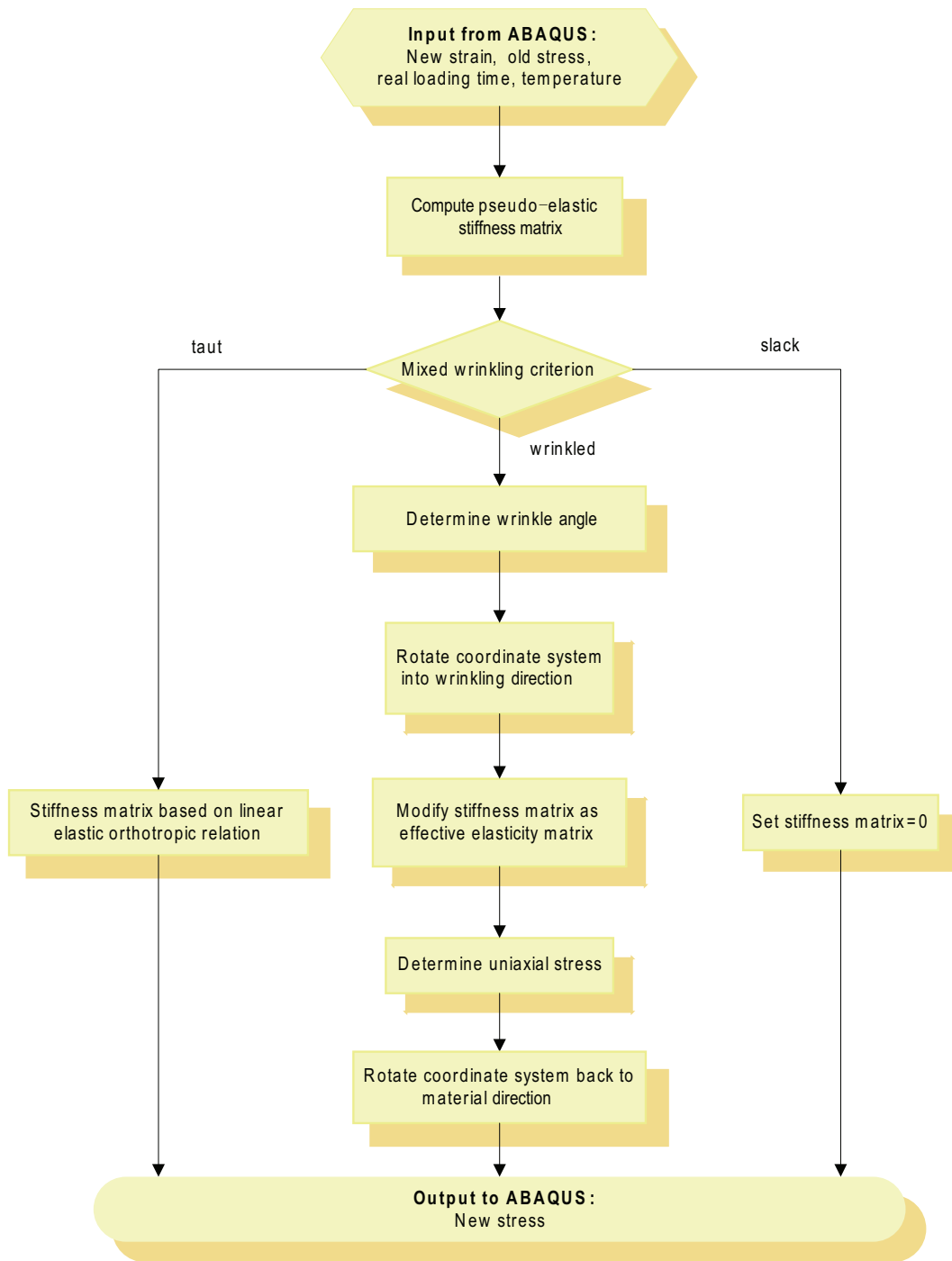
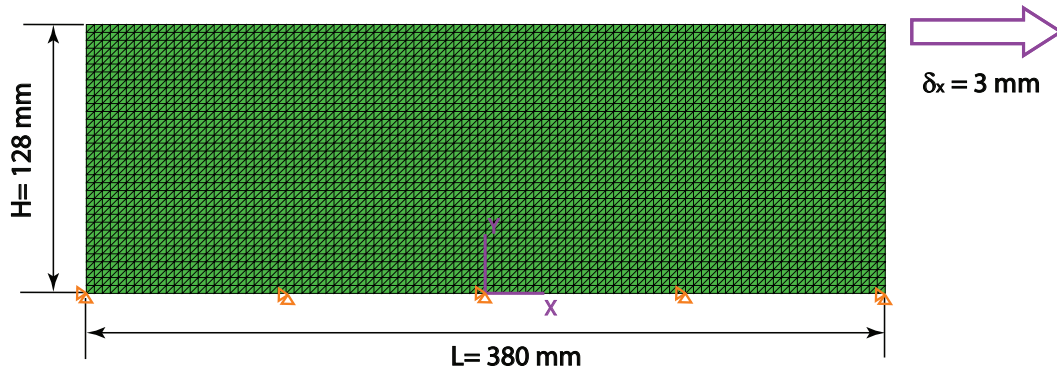


Figure 4. Wrinkling algorithm for linear orthotropic viscoelastic material subroutine VUMAT

**Table 3. Summary of membrane properties**

Length, $L$ (mm)	380
Height, $H$ (mm)	128
Thickness, $t$ (mm)	0.025
Young's Modulus, $E$ (N/mm <sup>2</sup> )	3530
Poisson's ratio, $\nu$	0.33
Density, $\rho$ (kg/mm <sup>3</sup> )	$1.46 \times 10^{-6}$



**Figure 5. Finite element mesh for membrane in shear.**

degrees of freedom on the upper and lower edges were constrained, to test the stability of the simulation.

## B. Results

The stress distribution corresponding to the final value of the horizontal displacement of the membrane is illustrated by means of contour plots and a vector plot, in Figure 6. Since the major principal stress,  $\sigma_1$ , is much greater than the minor principal stress,  $\sigma_2$ , only the major stress can be seen in the vector plot. The direction of the major principal stress, inclined at  $45^\circ$ , corresponds to the direction of the wrinkles. The dense distributions of vectors at the top-right and bottom-left corners indicate two areas of stress concentration.

The membrane finite element model using the VUMAT subroutine has succeeded in eliminating all negative stress, as illustrated by the minor stress distribution being approximately non-negative everywhere.

The principal stresses across the middle of the membrane are plotted in Figure 7.  $\sigma_1$  increases from zero at the edge to a uniform and positive value,  $\sigma_1 = 41.36$  MPa, whereas  $\sigma_2$  remains approximately zero. The major principal strain is at  $45^\circ$  and given by Wong and Pellegrino<sup>11</sup>

$$\epsilon_1 = \frac{\gamma}{2} \quad (39)$$

Since the shear strain is  $\gamma = \delta_x/H$ ,  $\sigma_1 = E\epsilon_1 = 41.37$  MPa. A detailed simulation with a thin-shell model of the membrane was carried out by Wong and Pellegrino<sup>12</sup> and the stress distribution obtained from that approach has been plotted for reference in Figure 7. **Figure 8 explains the wrinkling pattern for an isotropic membrane; it consists of a parallelogram region with a uniform wrinkling angle of  $45^\circ$  plus two triangular regions that are slack. Note that there are stress concentrations at the corners B and D whereas the stress is zero at the other two corners of the membrane, A and C.**

The energy variation during the Explicit simulation has been shown in Figure 9. The viscous dissipation is negligible (the linear viscosity coefficient was set equal to 0.005) and the kinetic energy is also small, indicating that a quasi-static simulation has been achieved. A constant strain energy level of 236.76 mJ during the second step indicates that the simulation is stable.

## VI. Experimental Studies of Viscoelastic Orthotropic Membranes

A rectangular membrane with exactly the same dimensions as the one considered in Section V.A and subject to the same loading condition has been tested. However, now the

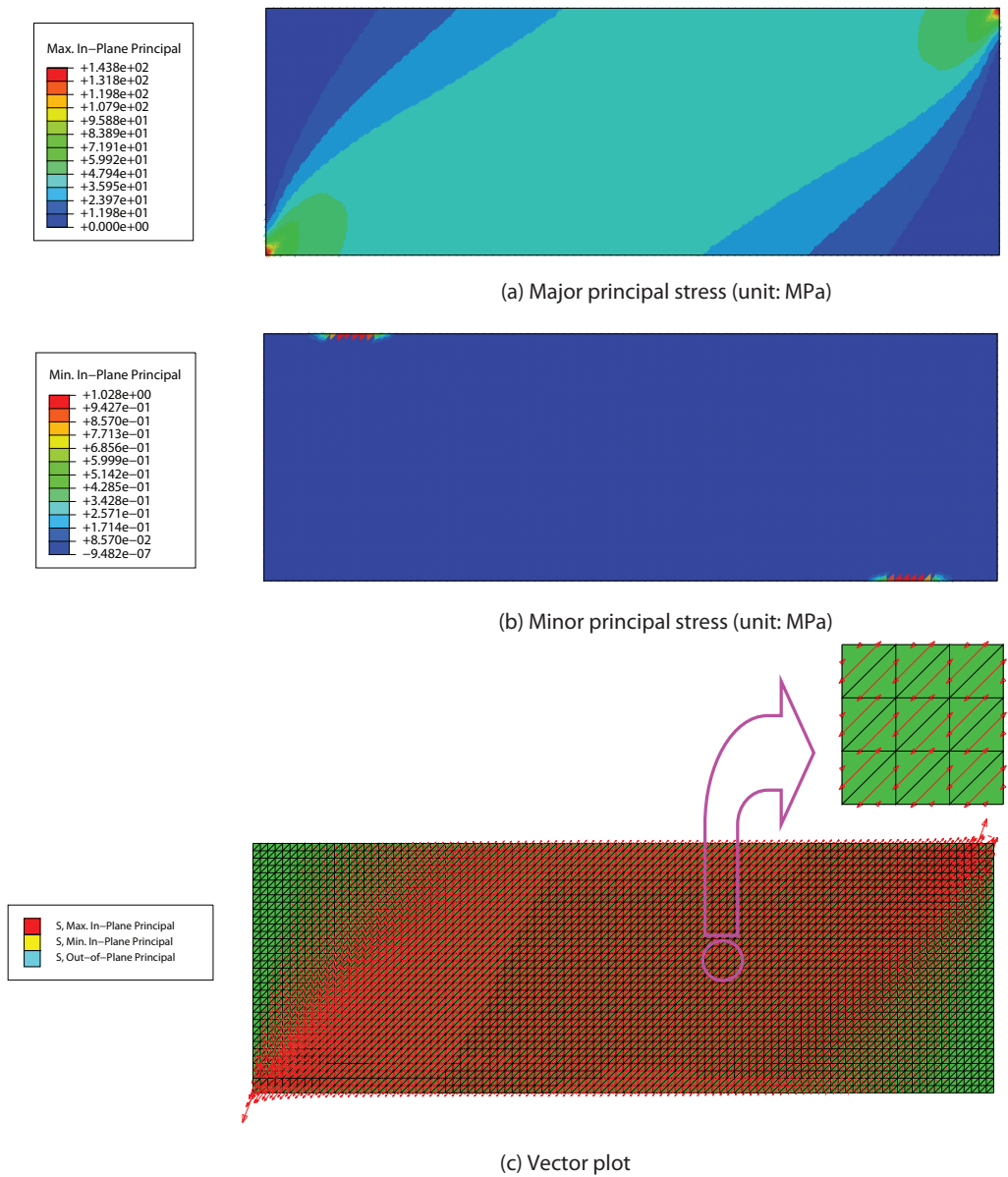


Figure 6. Plot of principal stress distribution for isotropic, time-independent membrane.

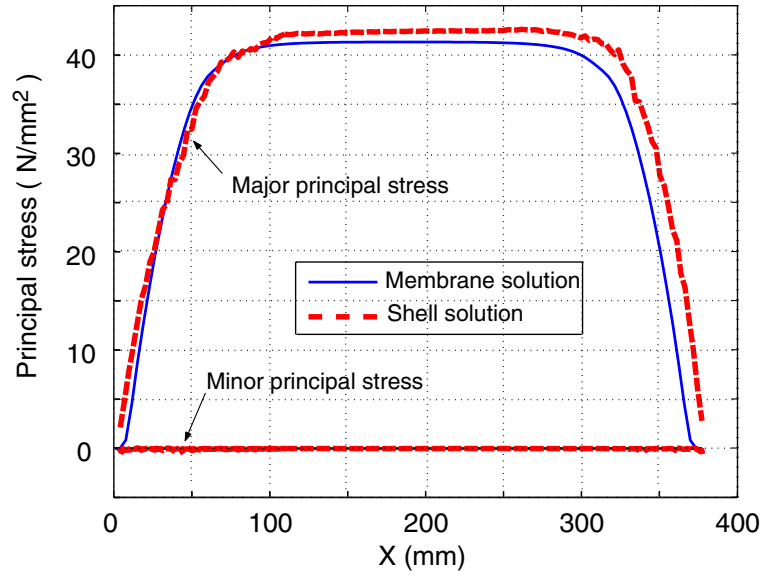


Figure 7. Variation of principal stresses across middle horizontal section of isotropic, time-independent membrane.

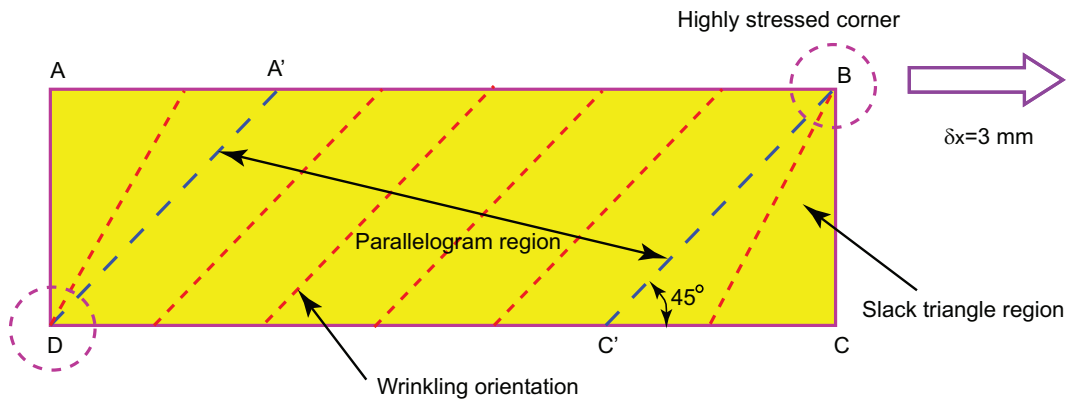
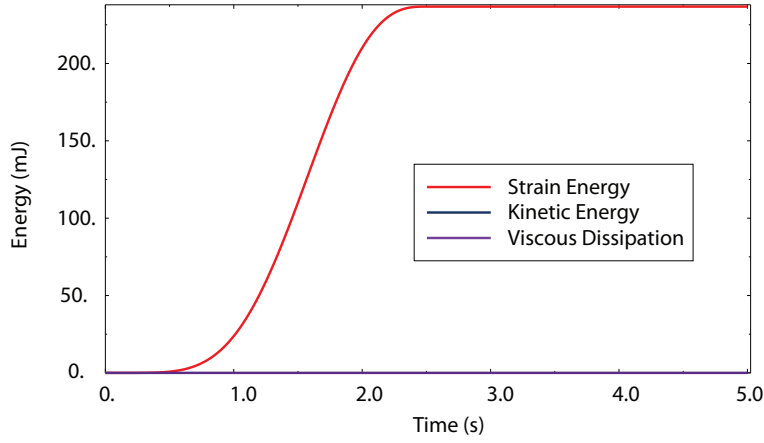


Figure 8. Wrinkling pattern for isotropic, time-independent membrane under simple shear.



**Figure 9. Energy variation for isotropic, time-independent membrane.**

situation is different in the following respects: (1) the membrane is StratoFilm 420, which is an orthotropic material showing nonlinear viscoelastic effects; (2) both loading and unloading are measured and hysteretic curves obtained from experiments will be compared with results of pseudo-elastic simulations; (3) after a linear ramp loading, the imposed displacement is kept constant for a certain time before unloading, to observe the viscoelastic response.

### A. Shear Test Rig

The scheme for the shear test was based on the concept previously adopted by Jenkins et al.<sup>30</sup> and then followed by Wong and Pellegrino.<sup>10</sup> The shear rig is shown in Figure 10, note that the moving edge block is attached to four linear bearings (LUI 5AL). A key step in the preparation for a test is setting up the flat membrane surface without any visible imperfections. To do this, a mist of distilled water was sprayed on a flat Perspex sheet and a piece of StratoFilm 420 with the required dimensions was put on the sheet. Then the film was lightly pressured with a hand roller to eliminate any air bubbles. The surface tension of the water at the interface holds the films against the plastic sheet. The upper and lower edges of one side of the film were glued to the fixed and moving edges of the rig, then clamping strips were glued and bolted to the other side of the film with M4 screws.

A controlled displacement of the moving edge block was imposed by means of a fine threaded screw, on the left-hand side of the rig and a force sensor (Futek LCM300) coaxial with the screw was used to measure the compressive contact force,  $F$ , between the screw and the moving edge block. A Keyence LK-G157 laser placed on the right-hand side (not shown in the figure) was used to measure the displacement  $\delta_x$  and the variation of force and displacement with time were measured with a Vishay System 7000 data logger. The scanning frequency was 128 samples per second during loading and unloading and 10 per

minute when the displacement was held constant. The data samples were averaged using a window of 20 data points to reduce noise. The repeatability of the force sensor is 0.1% of full range, i.e.  $\pm 1.11$  N; the resolution of the laser is  $0.5 \mu\text{m}$ .

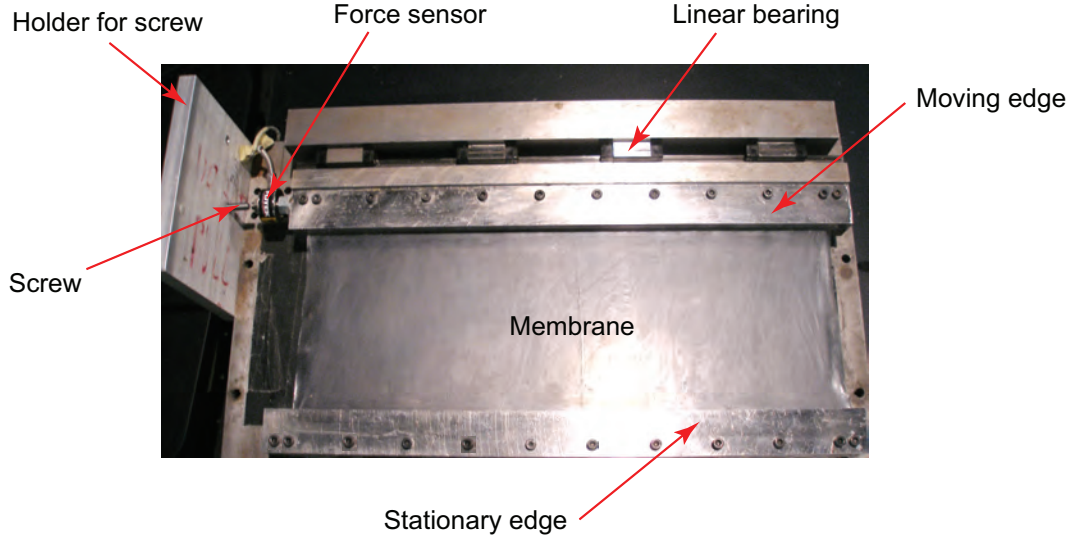


Figure 10. Shear rig.

## B. Experimental Procedure

StratoFilm 420 is orthotropic due to the non-uniform biaxial stretching during production. The directions of orthotropy are the machine direction, i.e. the direction in which the film is rolled, and the transverse direction. All experiments were carried out with the directions of orthotropy of the film parallel to the direction of shearing. In experiments denoted with an “M” the machine direction of the film was parallel to the direction of shearing, whereas in experiments denoted with a “T” the transverse direction of the film was parallel to the direction of shearing. For each set of experiments two displacement magnitudes were imposed, approximately 2 mm and 3 mm; in the second set of experiments the direction of motion was also reversed, after holding the displacement constant for a time  $t_C$ . The test parameters are summarized in Table 4.

It should be noted that at some point during the reversed direction of motion the reaction between the screw and the moving edge becomes tensile and at this point contact between the screw tip and the moving edge was lost. Hence, from this point on the motion of the moving edge was governed by the condition that the reaction force should be zero.

**Table 4. Test parameters**

Test No.	a	b	c	d
Direction of film	M	M	T	T
Displacement $\delta_x$ (mm)	2.31	3.36	2.39	3.34
Temperature $T$ (K)	294.55	294.65	294.65	294.65
Loading type	Loading only	Loading only	Loading & unloading	Loading & unloading
Linear ramp duration $t_L$ (s)	5.85	10.775	10.18	16.79
Constant displacement duration $t_C$ (s)	15145.70	13582.83	22115.49	21962.0
Linear ramp duration $t_U$ (s)	N.A.	N.A.	3.484	5.578

### C. Correction for Friction

The motion of the moving edge of the shear rig is resisted by friction in the linear bearings. It was assumed that the static and kinetic friction coefficients are equal and given by an expression of the type

$$f = f_0 + \mu P \quad (40)$$

where  $P$  is the total tension in the film, i.e. the total force in the direction perpendicular to the moving edge. The friction force  $f$  was measured on the shear rig without the film, by applying a known value of  $P$  and measuring the corresponding force needed to continuously move the sliding edge block.

The range of values of  $P$  that was considered was based on an ABAQUS simulation of the value corresponding to a film subject to a shear displacement  $\delta_x = 3$  mm. Hence three levels of  $P$  were applied: 0 N, 11.121 N and 22.241 N. Since the friction force  $f$  has much smaller values than the horizontal force, a dynamic force sensor with higher precision, PCB Piezotronics Model 208C01, was used to measure the friction force.

Two sets of test results have been plotted in Figure 11. The linear fit, in Newtons, is given by

$$f = +1.2976 + 0.00421 P \quad (41)$$

For simplicity the constant value  $f = 1.298$  N was assumed, as it turned out that  $P$  is less than 22.241 N in all tests. In conclusion, the shear force acting on the film,  $S$ , can be obtained by correcting the total applied force  $F$ , measured by the force sensor during the



test, see Figure 12 for details:

$$S = F - f \text{ if } \dot{\delta}_x > 0 \quad (42)$$

$$S = F + f \text{ if } \dot{\delta}_x < 0 \quad (43)$$

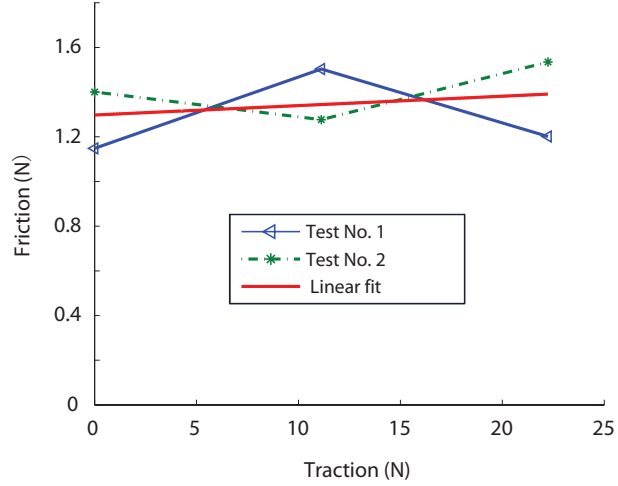


Figure 11. Fitting of friction force.

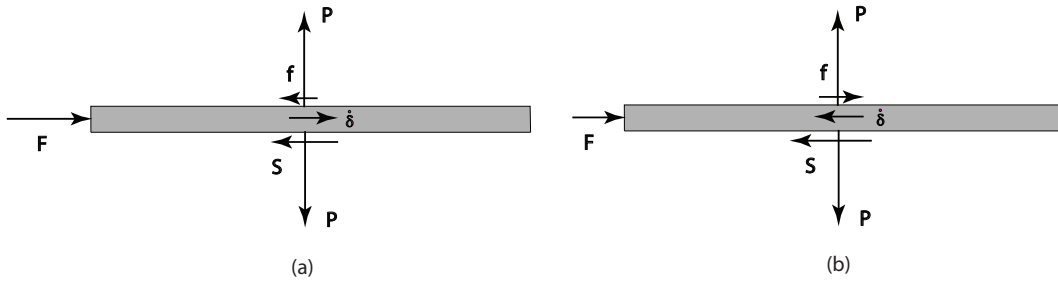
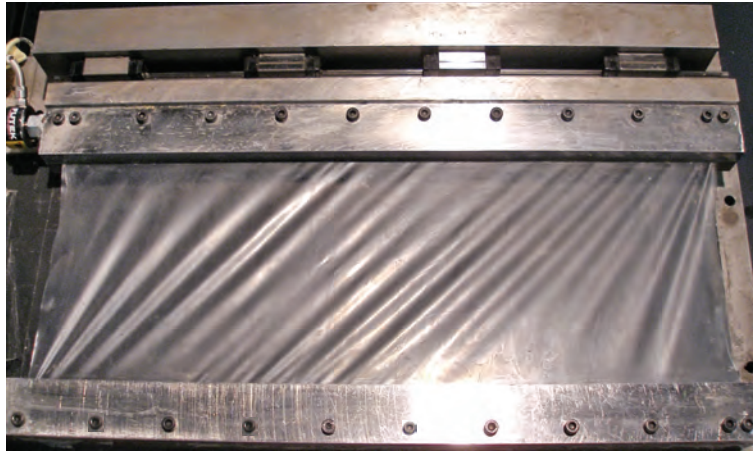


Figure 12. Free bodies of sliding edge block to derive correction of shear force.

#### D. Results of Shear Tests

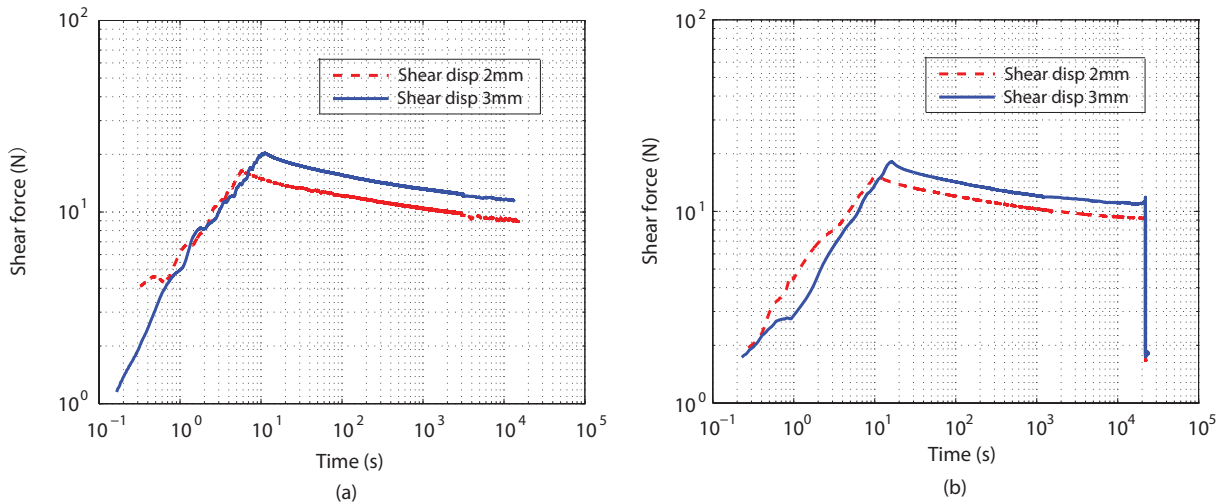
The wrinkling pattern observed on a film subject to a 3 mm shear displacement is shown in Figure 13. The wrinkles are almost evenly distributed in the central parallelogram region. The top left and bottom right corner regions are slack and so there are no wrinkles in these regions.

A correction to the measured total force has been applied, using Equations 42 and 43 with a constant friction value, have been applied to correct the shear force, respectively for the cases of loading and unloading. The results have been plotted in Figures 14 and 15.



**Figure 13.** Wrinkling pattern for 38 micron thick Stratofilm 420 subject to 3 mm shear displacement.

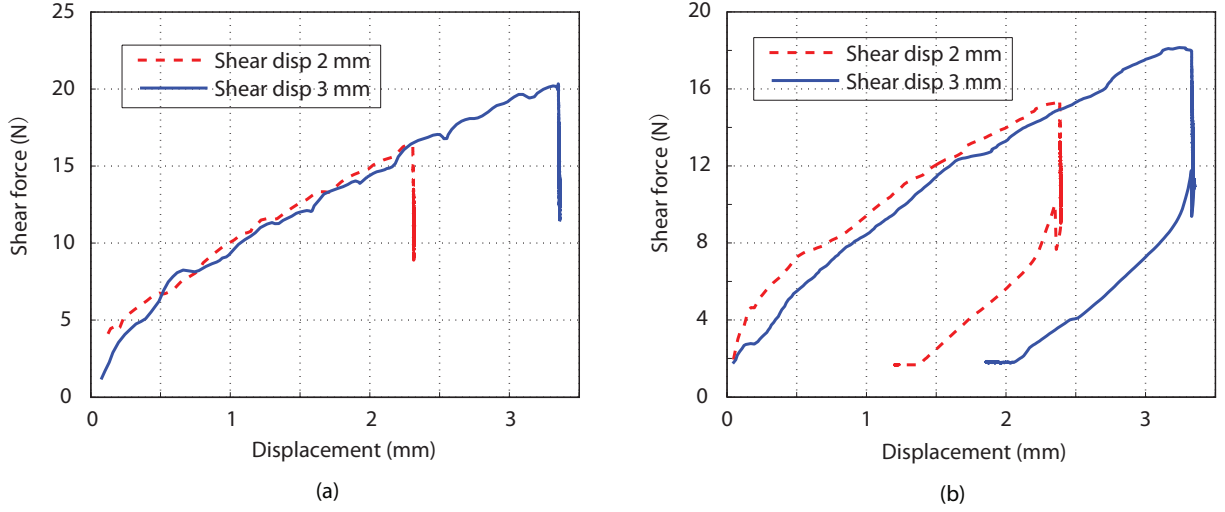
A comparison between the shear force time variation between the four tests is shown in Figure 14. The increase in shear force is approximately linear during the loading ramp. Then it gradually decreases when the displacement remains constant and tends to a constant value.



**Figure 14.** Measured time dependence of shear force, after correction for friction effects: (a) “M” machine direction parallel to direction of shearing; (b) “T” transverse direction parallel to direction of shearing.

The relationship between the shear force and displacement for the four tests is shown in Figure 15. The curves for total shear displacements of 2 mm and 3 mm generally follow the same shape and kinks appear at the point of initial unloading due to the shear force correction when the friction suddenly changes direction. Clearly the simplified friction correction model

that **has** been used does not do a good job at the point of transition.



**Figure 15.** Measured force-displacement relationships, after correction for friction effects: (a) “M” machine direction parallel to direction of shearing; (b) “T” transverse direction parallel to direction of shearing.

Since the differences in corresponding displacements are minor, the temperature did not change and the loading can be assumed quasi-static, it is interesting to compare the maximum shear forces in the four tests, see Table 5. With everything else fixed, one finds that the greater the shear displacement, the greater the shear force is. Also, for the same displacement, the film arranged with the machine direction parallel to the long edge carries a higher shear force than the film placed in the transverse direction; however the difference is generally less than 10%. This is because Stratofilm 420 has only a weak anisotropy.

**Table 5.** Maximum shear forces

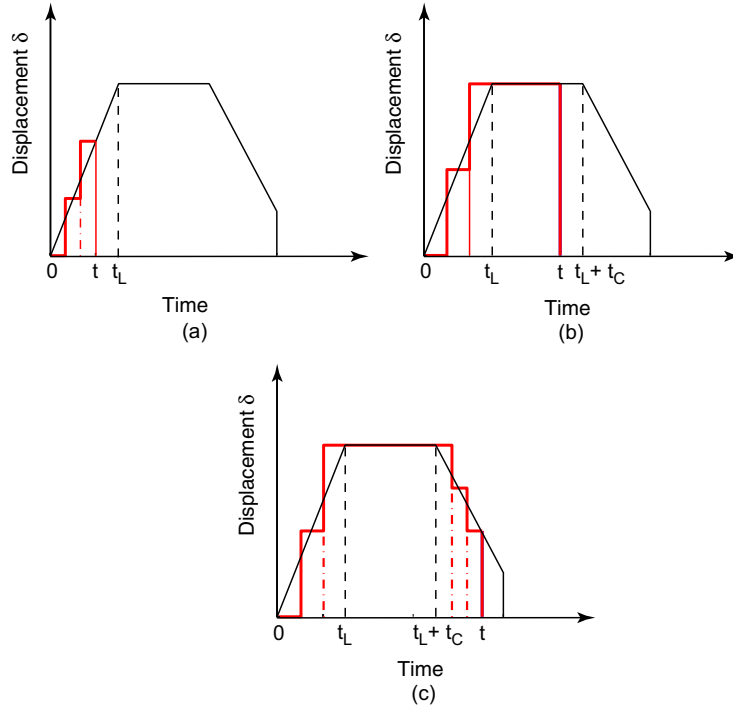
Test No.	a	b	c	d
Direction of film	M	M	T	T
Displacement $\delta_x$ (mm)	2.31	3.36	2.39	3.34
Max total horizontal force $F$ (N)	17.618	21.206	16.633	19.271
Max shear force $S$ (N)	16.320	19.908	15.335	17.973

## VII. Comparison of Results and Discussion

The general orthotropic viscoelastic membrane model can be tested by comparing the simulation results to the results from the experiments. The first issue is how many steps

should be used to represent the time history of the imposed displacement. As an example, we have investigated the linear loading ramp for the M test with a displacement  $\delta_x = 3.36$  mm, at the time  $t_L = 10.775$  s. We have found that the total estimate for the shear force at this time increases from 16.445 N, to 18.970 N, and 20.117 N for single-step, two-step and three-step approximations, respectively. The first two results differ by 15% but the difference of the last two reduces to 6%. Taking 20.117 N as the correct solution, we have assumed that a two-step approximation for linear ramp loading is sufficiently accurate. The experimental measurement of the force at this time was 19.908 N, which confirm that the two-step solution is close to the physically correct value. **At the stress-concentration corners of the membrane there is a maximum principal strain of 5.2%, which is at the limit of Rand's viscoelastic model.<sup>25,26</sup>**

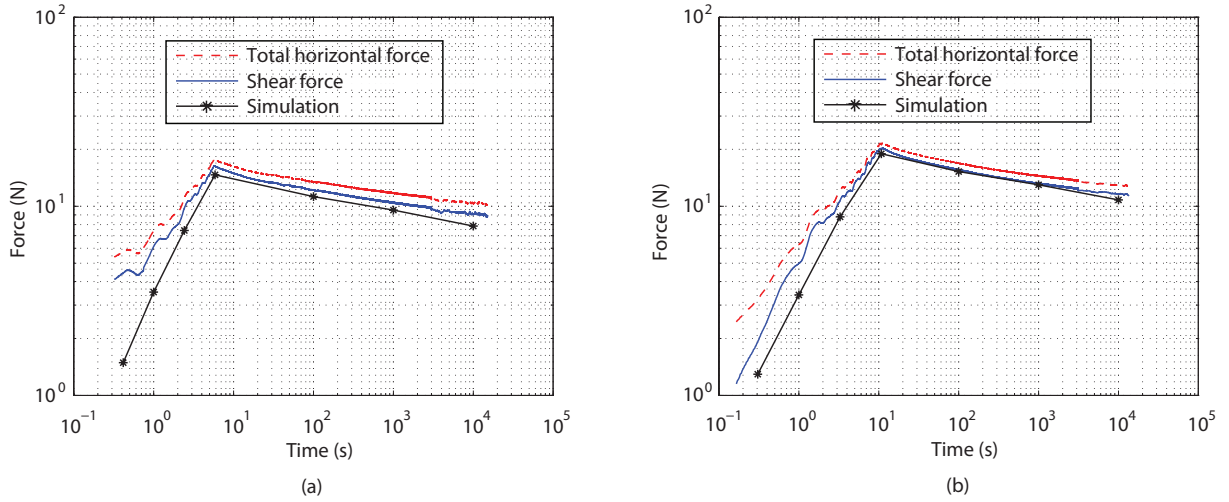
Based on this observation, the discretization technique that has been adopted for the actual displacement function is shown in Figure 16, depending on where the chosen time  $t$  falls with respect to the ramp times.



**Figure 16.** Discretization of displacement function using two-steps for each ramp; (a) case  $t < t_L$ ; (b) case  $t_L < t < t_L + t_C$ ; (c) case  $t_L + t_C < t < t_L + t_C + t_U$ .

The first thing to note from the simulation results is that the wrinkling angles are on average  $44.6^\circ$  and  $48.3^\circ$ , respectively, in the M and T cases. This difference is too small to be detected in the experiments. Figures 17 and 18 compare the relationships between total shear force and time obtained from the simulations with both uncorrected and friction corrected

experimental results. During the initial loading, especially the first 1 s, the comparison shows significant differences; however this is not surprising. The differences are due to the presence of inertia forces in the experimental results, due to the mass of the edge block that was not included in the finite-element model, and to the fact that when the shear forces are small the force sensor is less accurate and also friction is comparatively large and so our rather rough technique for removing friction effects is not accurate. The simulations and the experimental results are much closer from  $t \approx 2$  s onwards and particularly for the case  $\delta_x = 3$  mm. In Figure 17(b) during the constant displacement phase the maximum difference between simulation and friction-corrected measured shear forces is 0.83 N, or 7.15%. In Figure 18(b1), the maximum difference the maximum difference is 0.58 N, or 5.28%. Figure 18(b2) provides a more detailed comparison of the unloading curves shown in Figure 18(b1), to provide a more detailed comparison of the unloading curves. In this range the maximum difference is 0.74 N.



**Figure 17.** Comparison between simulation and measurement for “M” test: (a) maximum shear displacement 2 mm; (b) maximum shear displacement 3 mm.

The hysteretic relationship between shear force and displacement is shown in Figure 19. Different loading and unloading paths result from the cumulative effects of viscoelasticity, which has a significant influence on the behavior of the wrinkled films.

## VIII. Conclusion

The objective of this study was to investigate the wrinkling of orthotropic viscoelastic membranes. The method developed in this paper, based on a modification of the material stiffness matrix to incorporate the effects of orthotropic wrinkling and viscoelasticity, has been shown to be an effective way of capturing in a finite element simulation the behavior

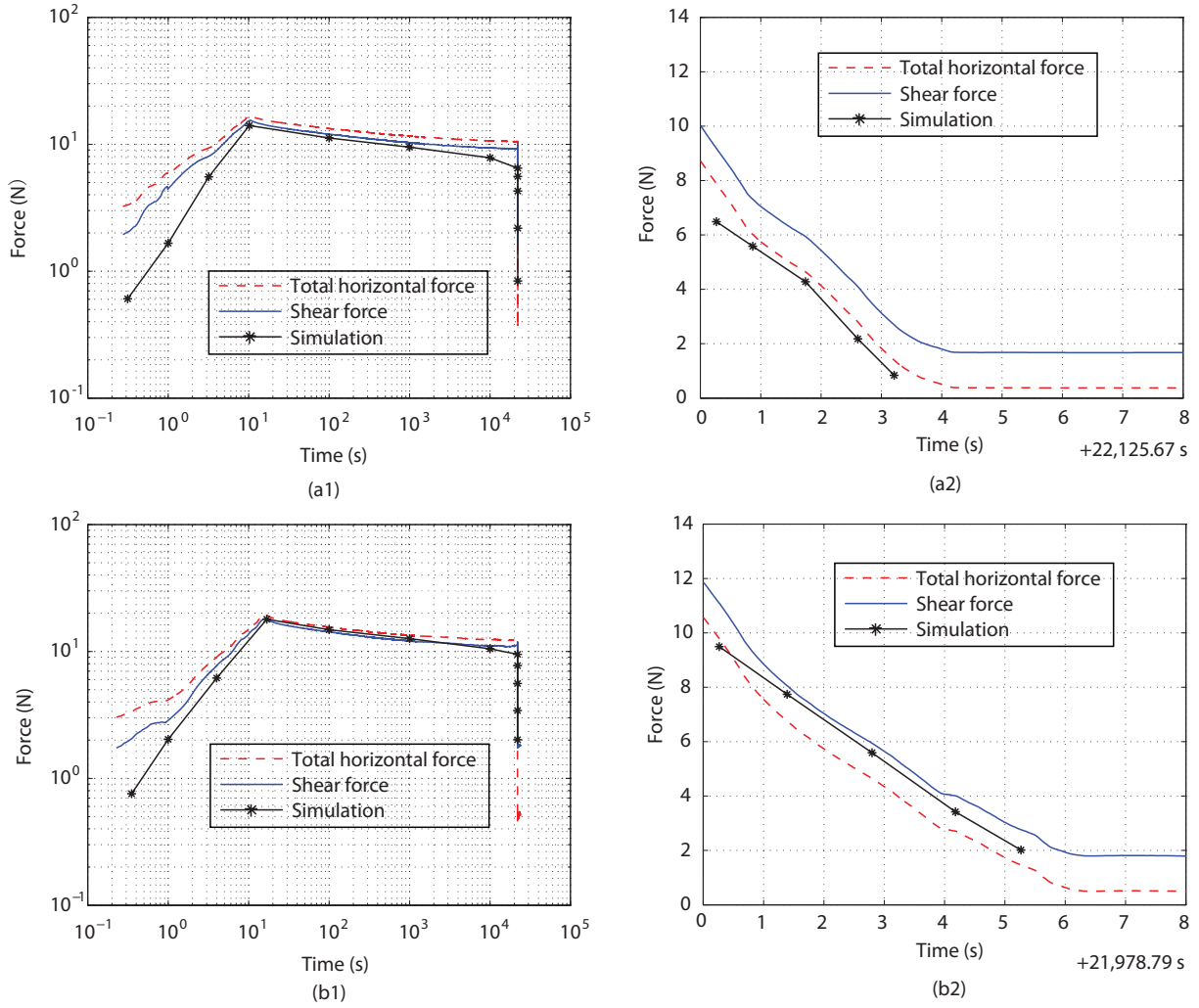
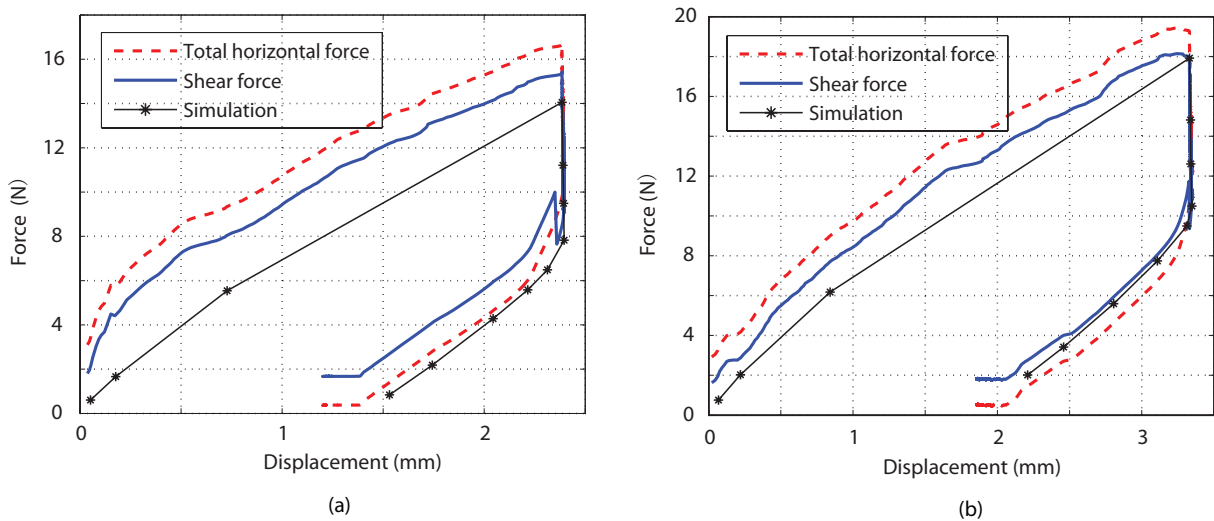


Figure 18. Comparison between simulation and experiment for “T” test: (a1) full behavior for maximum shear displacement 2 mm; (a2) unloading curve for maximum shear displacement 2 mm; (b1) full behavior for maximum shear displacement 3 mm; (b2) unloading curve for maximum shear displacement 3 mm.



**Figure 19.** Comparison between simulation and experiment for “T” test: (a) maximum shear displacement 2 mm; (b) maximum shear displacement 3 mm.

of a real plastic film.

In the wrinkling model the state of a membrane element (taut, wrinkled, slack) is defined by a mixed criterion. Once it has been established that an element is wrinkled, an iterative scheme searches for the wrinkle orientation angle and the correct stress distribution, involving only uniaxial tension in the wrinkle direction, is then obtained.

This wrinkling model has been validated by a comparison with a published solution for the case of a time-independent isotropic membrane under simple shear. The model has then been applied to a particular type of viscoelastic balloon film, known as StratoFilm 420. Using the Rand-Schapery model for this film, a time-dependent pseudo-elastic stiffness matrix was defined and so, instead of having to compute the convolution integral throughout the simulation, we were able to compute the behavior of a membrane structure by superposition of a series of incremental response functions. These calculations were implemented in standard finite element software.

Experimental tests on StratoFilm 420 under simple shear were carried out, including loading and unloading, and the experimental results were compared with ABAQUS/Explicit simulations. The results agree very well once friction and dynamic effects are excluded. There are differences of about 10% in the shear forces measured on membranes arranged in different material directions because StratoFilm 420 has only weak anisotropy; hence the direction of the wrinkles changed by less than  $4^\circ$  when the film was rotated through  $90^\circ$ . Significant differences were seen between loading and unloading curves, indicating that viscoelastic behavior is a significant source of energy dissipation.

Viscoelastic simulations based on the convolution integral in Eq. 7 would be computa-

tionally intensive for large structures undergoing significant geometry changes, such as the deployment of balloons. Moreover, computing intermediate results at many stages of deployment of a balloon is of little interest as the stresses remain small until the balloon reaches its final configuration and begins to pressurize. This paper has shown that for the balloon film SF 420 acceptable results are obtained from pseudo-elastic simulations based on a two-step representation of each ramp in the loading function. Therefore, applications of the present approach to balloon simulations is expected to be much more efficient than attempting to predict the detailed behavior of the balloon using a much finer representation of the loading together with an incremental form of the convolution integral.

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