

Phase-Field theory and modeling of dislocations in ductile single crystals

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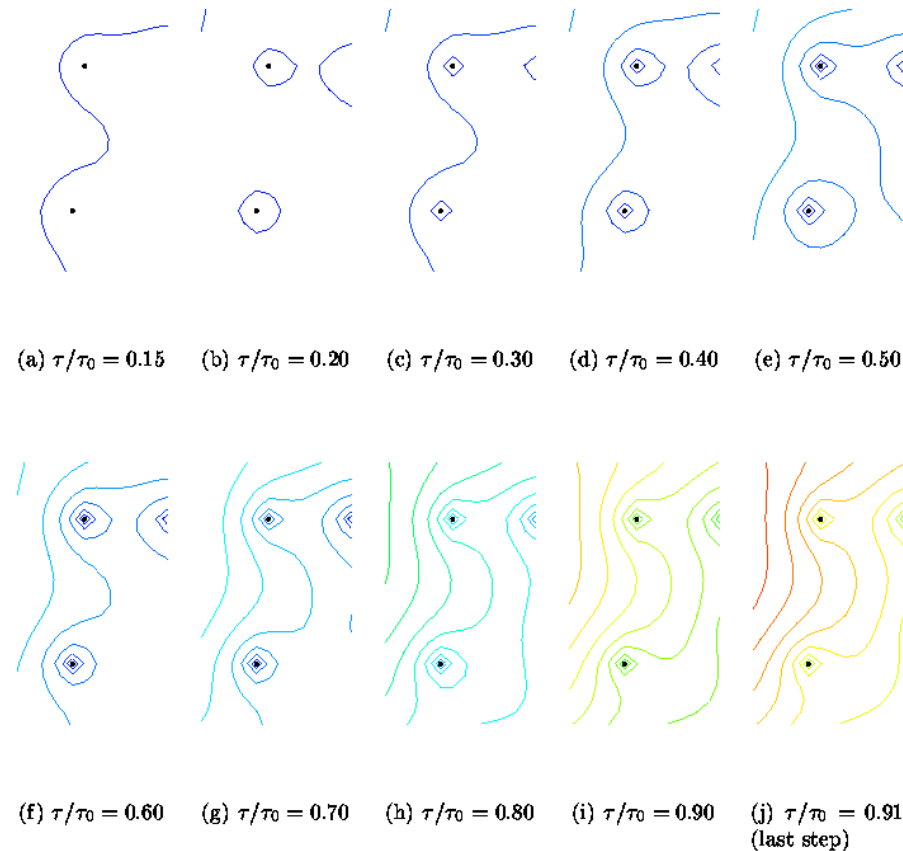
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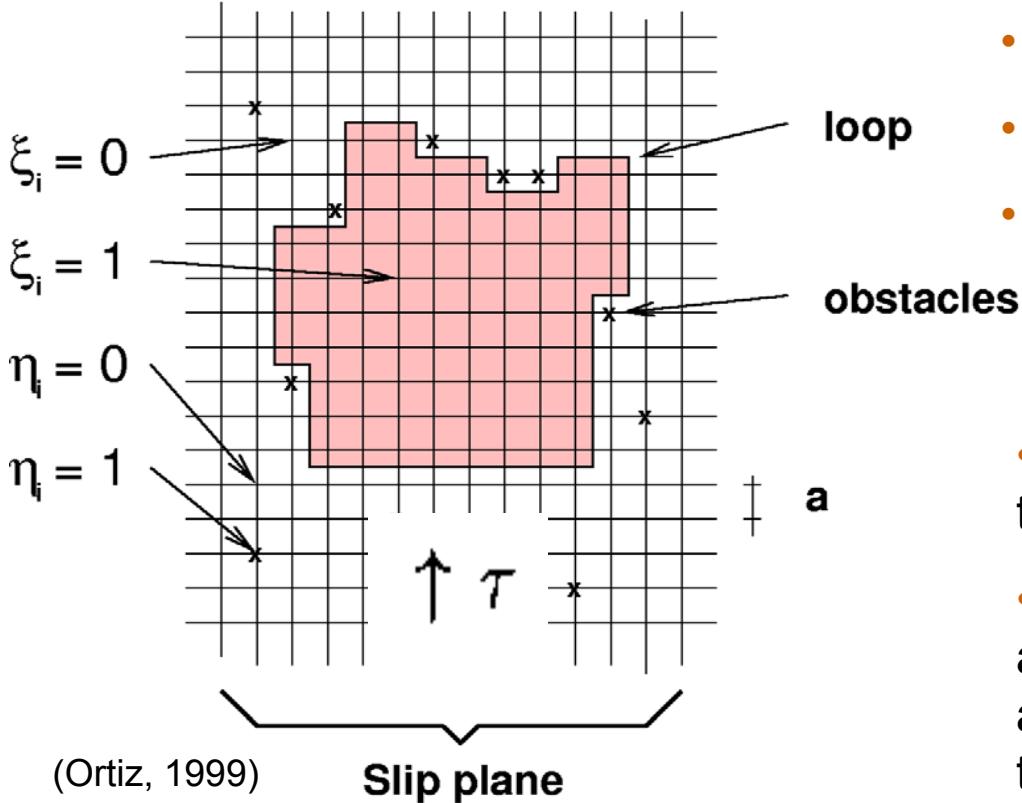


Motivation

- The aim of this study is to develop a **phase-field theory of dislocation dynamics**, strain hardening and hysteresis in ductile single crystals.
- This representation enables to identify individual dislocation lines and arbitrary dislocation geometries, including tracking intricate topological transitions such as loop nucleation, pinching and the formation of **Orowan loops**.
- This theory permits the **coupling between slip systems**, consideration of obstacles of varying strength and **anisotropy** and **thermal effects**.
- **Rate effects** are observed at finite temperatures.



Overview



Lattice model of dislocation
loop-point obstacle interaction

Effective Dislocation Energy

- Core Energy
- Dislocation Interaction
- Irreversible Obstacle Interaction

Equilibrium configurations

- Closed form solution at zero temperature.
- Metropolis Monte Carlo algorithm and mean field approximation at finite temperatures.

Macroscopic Averages

$$\gamma \propto \langle \xi \rangle \quad \rho \propto \langle |\nabla \xi| \rangle$$



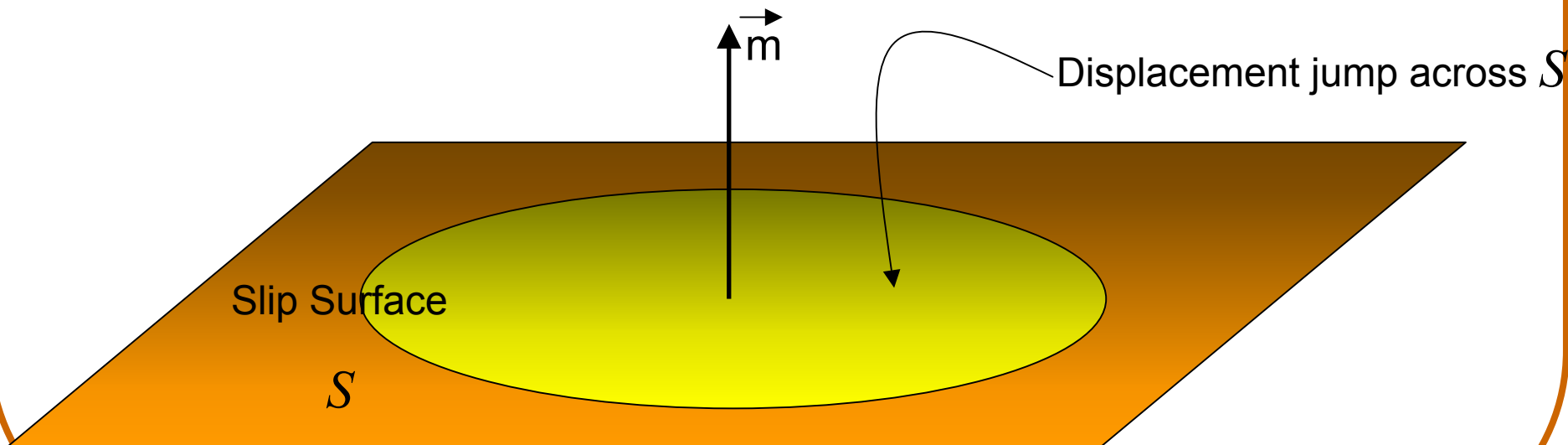
Effective Energy

$$E[u] = \underbrace{\frac{1}{2} \int c_{ijkl} \beta_{ij}^e \beta_{kl}^e d^3x}_{\text{Elastic interaction}} + \underbrace{\int_S \phi(\delta) dS}_{\text{Core energy}} - \underbrace{\int_S t_i \delta_i dS}_{\text{External field}}$$

where

$$u_{i,j} = \beta_{ij}^e + \beta_{ij}^p$$

with $\beta_{ij}^p = \delta_i m_j \delta_D(S)$



Elastic interaction

$$E^{\text{int}} = \frac{1}{2} \int c_{ijkl} \beta_{ij}^e \beta_{kl}^e d^3x$$

Elastic distortion:

$$\beta_{kl}^e = -G_{ki,l} * \left(c_{ijmn} \beta_{mn}^p \right)_{,j} - \beta_{kl}^p$$

$$E^{\text{int}}[\zeta] = \frac{1}{(2\pi)^2} \int \frac{\mu b^2}{4} K(k_1, k_2) |\hat{\zeta}|^2 d^2k$$

with $\zeta(x_1, x_2) = \delta(x_1, x_2) / b$

$$K(k_1, k_2) = \frac{1}{1-\nu} \frac{k_1^2}{\sqrt{k_1^2 + k_2^2}} + \frac{k_2^2}{\sqrt{k_1^2 + k_2^2}}$$

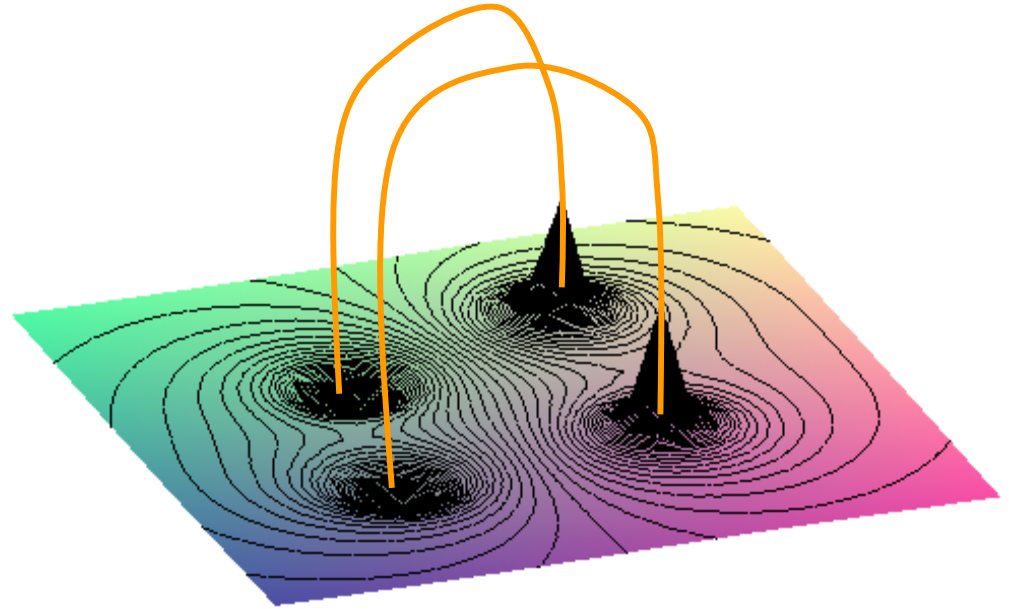


External Field

$$E^{ext} = \int_S \delta_i t_i dS$$

with: $\delta_i(x_1, x_2) = \zeta(x_1, x_2) b_i$

$$s(x_1, x_2) = t_i(x_1, x_2) b_i$$



$$s(x) = \tau + \sum_{n=1}^{Ndis} \frac{C_n}{|x - x_n|}$$

applied
shear stress

forest dislocations

$$E^{ext} = \int_S b s \zeta dS$$



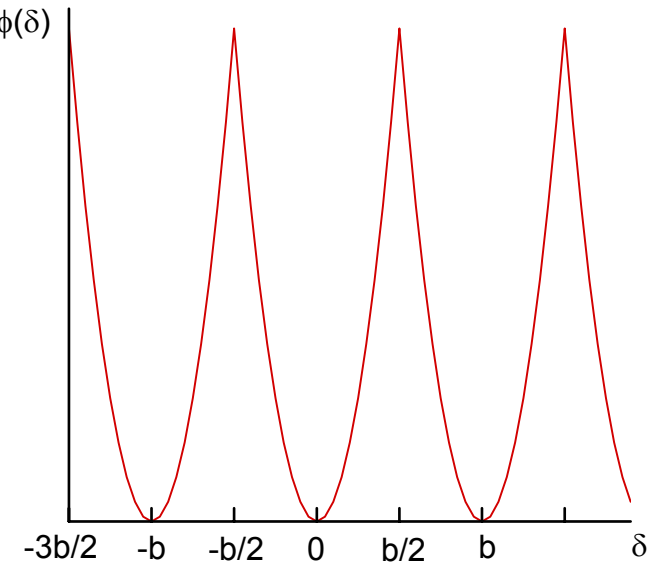
Core Energy

$$\phi(\delta) = \min_{\xi \in Z} \frac{1}{2} C_{ij} (\delta_i - \xi b_i) (\delta_j - \xi b_j)$$

$$\xi : S \rightarrow Z \quad C_{ij} = \frac{\mu}{2d} \delta_{ij}$$



$$\phi(\zeta) = \inf_{\xi \in Z} \frac{b^2 \mu}{4d} |\zeta - \xi|^2$$



Ortiz and Phillips, 1999

$$E^{core} = \int_S \phi(\zeta) dS$$



Phase-Field Energy

$$E[\xi] = \inf_{\zeta \in Y} \frac{1}{(2\pi)^2} \int \left(\frac{\mu b^2}{2d} |\hat{\zeta} - \hat{\xi}|^2 + \frac{\mu b^2}{4} K |\hat{\zeta}|^2 - b \hat{s}^* \hat{\zeta} \right) d^2 k$$

Minimization with respect to ζ gives:

$$\hat{\zeta} = \frac{d}{\mu b} \frac{\hat{s}}{1 + Kd/2} + \frac{\hat{\xi}}{1 + Kd/2}$$

core regularization factor

$$E[\xi] = \frac{1}{(2\pi)^2} \left[\int \frac{\mu b^2}{4} \frac{K}{1 + Kd/2} |\hat{\xi}|^2 d^2 k - \int \frac{b \hat{s}^* \hat{\xi}}{1 + Kd/2} d^2 k \right] + E_0$$

elastic energy



Phase-Field Energy

Elastic energy

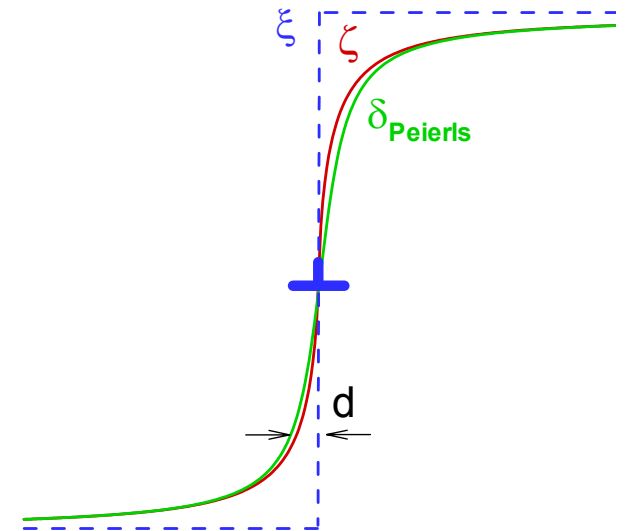
$$E_0 = \frac{1}{(2\pi)^2} \int \frac{d}{2\mu} \frac{|\hat{s}|^2}{1 + K d/2} d^2 k$$

Core regularization

$$\zeta = \zeta_0 + \varphi_d * \xi$$

Core regularization factor

$$\hat{\varphi}_d(k) = \frac{1}{1 + K d/2}$$



$$\delta_{\text{Peierls}} = -\frac{b}{2\pi} \tan^{-1} \frac{2(1-\nu)x}{d}$$



Irreversible Process and Kinetics

- Irreversible dislocation-obstacle interaction may be built into a variational framework, we introduce the incremental work function:

$$W[\xi^{n+1} | \xi^n] = E[\xi^{n+1}] - E[\xi^n] + \underbrace{\int f(x) |\xi^{n+1} - \xi^n| d^2x}_{\text{incremental work dissipated at the obstacles}}$$

incremental work dissipated at the obstacles

- Primary and forest dislocations react to form a jog: $f \propto \frac{\mu b^2}{4\pi}$
- Updated phase-field follows from: $\inf_{\xi_{n+1} \in X} W[\xi^{n+1} | \xi^n]$
- Short range obstacles:
$$f(x) = b\tau^P + \sum_{i=1}^N f_i \varphi_d(x - x_i)$$



Irreversible Process and Kinetics

$$\int f(x) |\zeta^{n+1} - \zeta^n| d^2x = \sup_{|g^{n+1}| \leq f} \int g^{n+1}(x) (\zeta^{n+1} - \zeta^n) d^2x$$

Kuhn-Tucker optimality conditions:

$$\xi_i^{n+1} - \xi_i^n = \lambda_i^+ - \lambda_i^-$$

$$g_i^{n+1} - f_i \leq 0$$

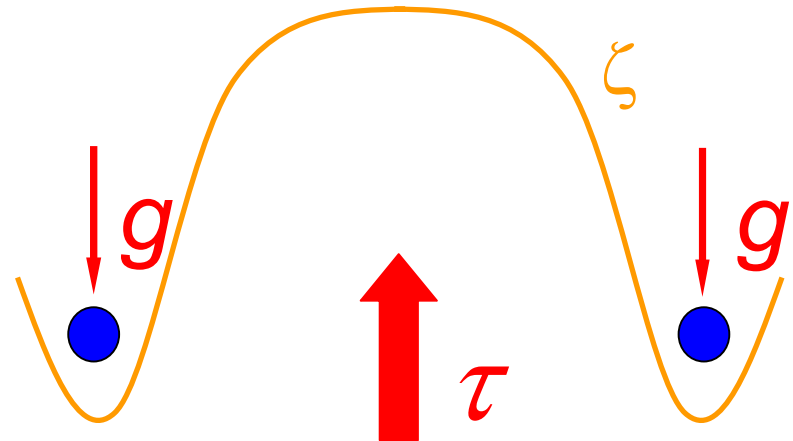
$$-g_i^{n+1} - f_i \leq 0$$

$$\lambda_i^+ \geq 0$$

$$\lambda_i^- \geq 0$$

$$(g_i^{n+1} - f_i) \lambda_i^+ = 0$$

$$(g_i^{n+1} + f_i) \lambda_i^- = 0$$



Equilibrium condition:

$$b \tau^{n+1} = \frac{1}{\Omega} \sum_{i=1}^N g_i^{n+1}$$



Energy minimizing phase-field

Unconstrained minimization problem: $\inf_{\eta \in Y} W[\eta] \longrightarrow \frac{\mu b}{2} K \hat{\eta} = \hat{s}$

if $\langle s \rangle = 0$

$$\eta = G * s + C$$

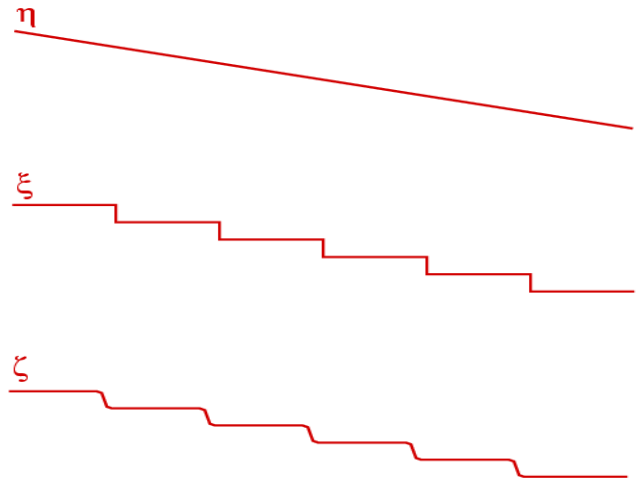
with

$$G(x) = \frac{2}{\mu b} \int \frac{e^{ik \cdot x}}{(2\pi)^2 K} d^2 k = \frac{1}{(\mu b \pi)} \frac{\sqrt{x_1^2 + x_2^2}}{x_1^2 + x_2^2 / (1 - \nu)}$$

$$\zeta = \varphi_d * P_X(G * s) + C$$

P_X

φ_d

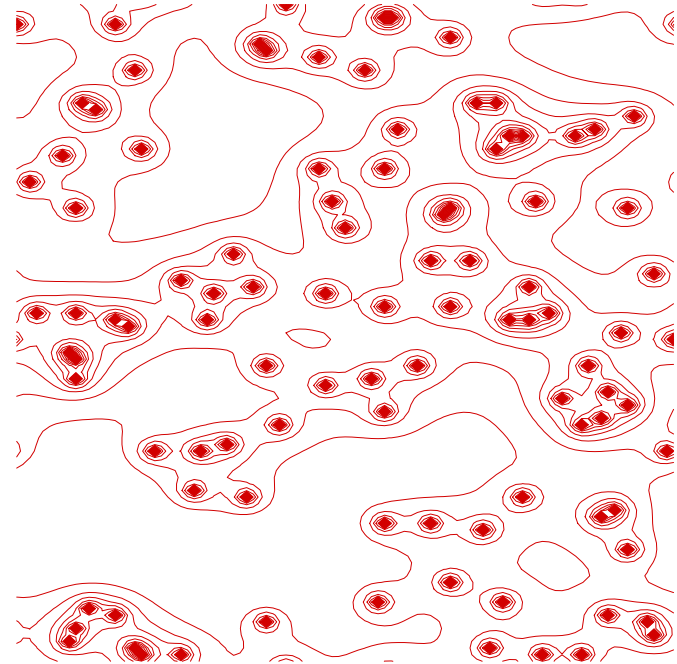


Closed-form solution

$$\eta_i^{n+1} = -\sum_{j=1}^N G_{ij} g_j^{n+1} + C^{n+1}$$

$$G_{ij} = G(x_i - x_j)$$

$$G_{ii} = \frac{(2-\nu)\sqrt{1-\nu}}{2\pi^2 b\mu} \frac{1}{d}$$



Dislocation loops

Macroscopic averages

- Slip

$$\gamma = \frac{b}{lN} \sum_{i=1}^N \xi_i = \gamma_0 \langle \xi \rangle \quad l \approx 10^3 b$$

- Dislocation density

$$\rho = \rho_0 b \langle |\nabla \xi| \rangle \quad \rho_0 = \frac{1}{bl} \approx 10^{16} \frac{1}{m^2}$$

- Obstacle concentration

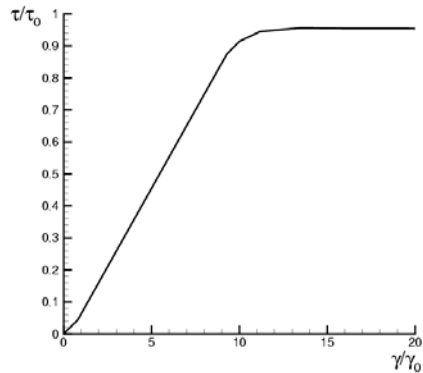
$$c = \langle \eta \rangle \quad c \approx 10^{12} - 10^{17} \frac{1}{m^2}$$

- Shear stress

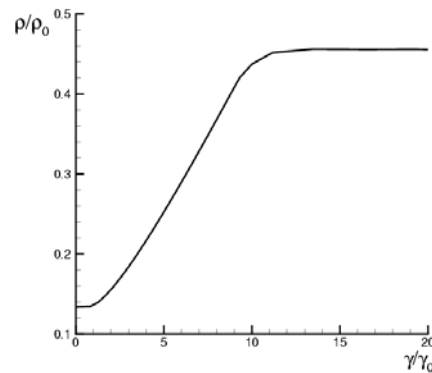
$$\tau = \tau_0 \frac{\langle f \rangle}{\langle F^{obs} \rangle} \quad \tau_0 = F^{obs} c \approx 10^{-2} - 10^{-4} \mu$$



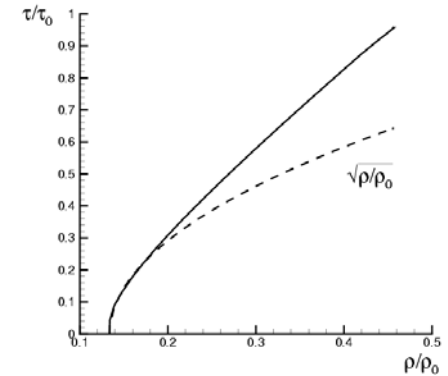
Monotonic loading



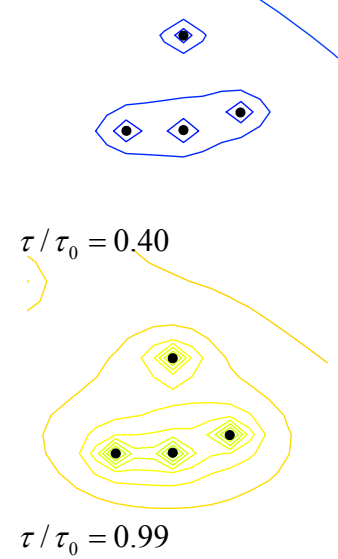
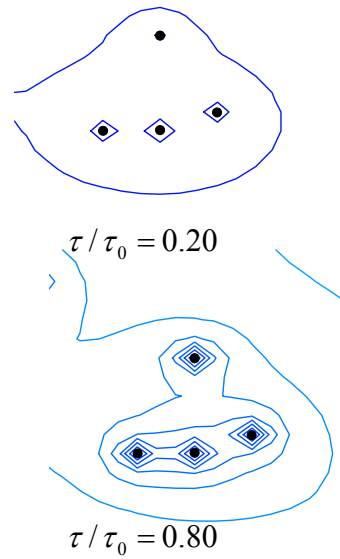
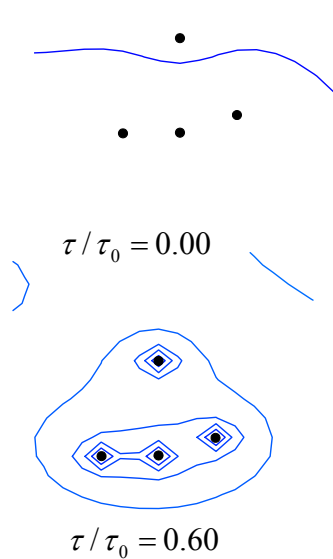
Stress-strain curve.



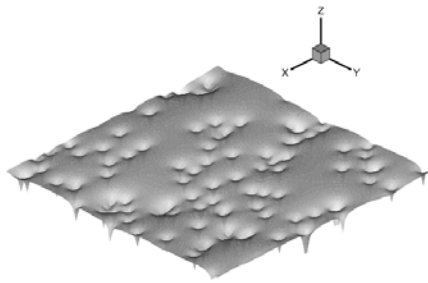
Evolution of dislocation density with strain.



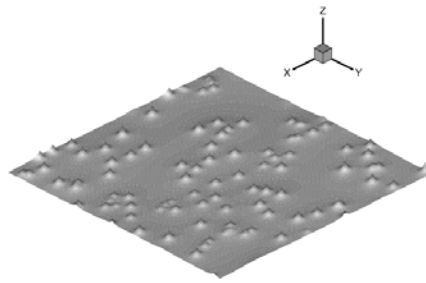
Evolution of dislocation density with shear stress.



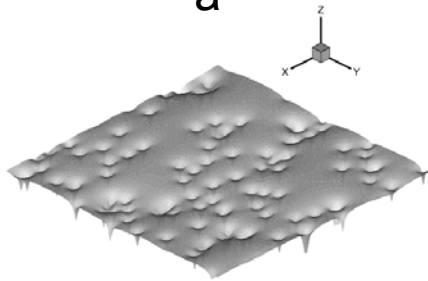
Fading memory



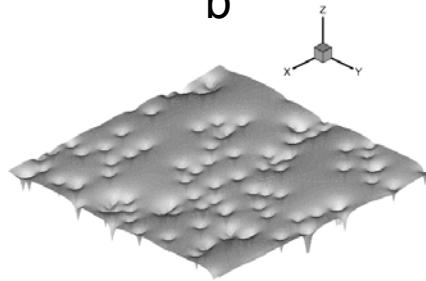
a



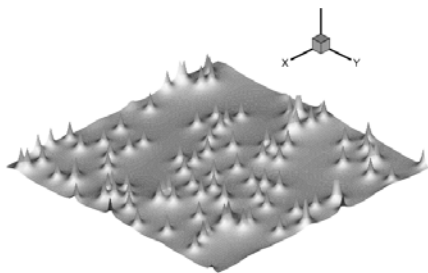
b



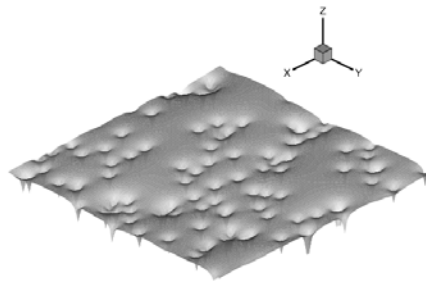
c



d

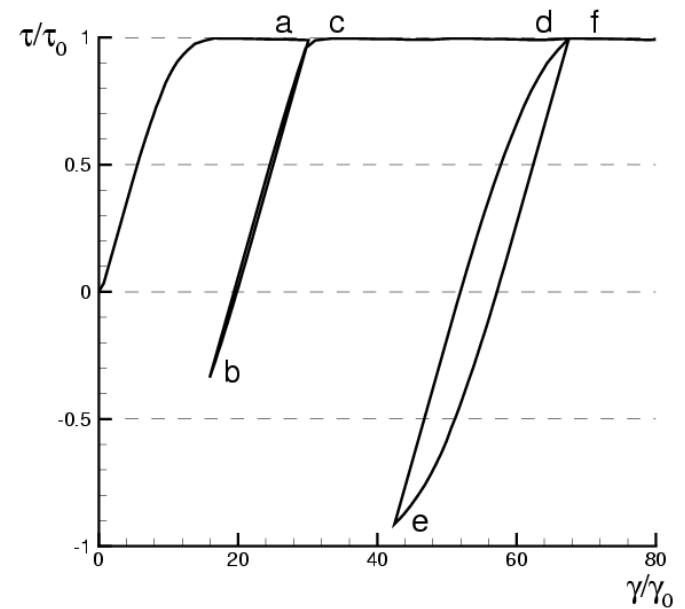


e



f

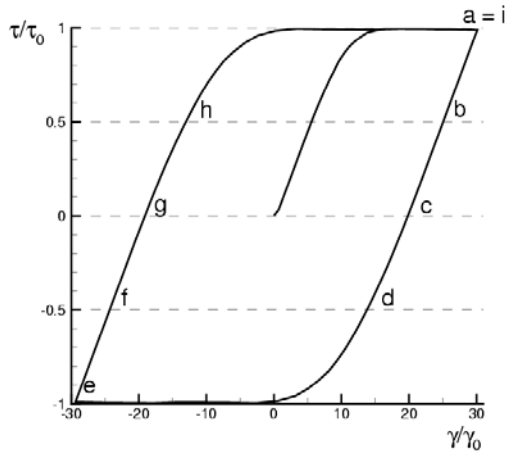
Three dimensional view of the evolution of the phase-field, showing the the switching of the cusps.



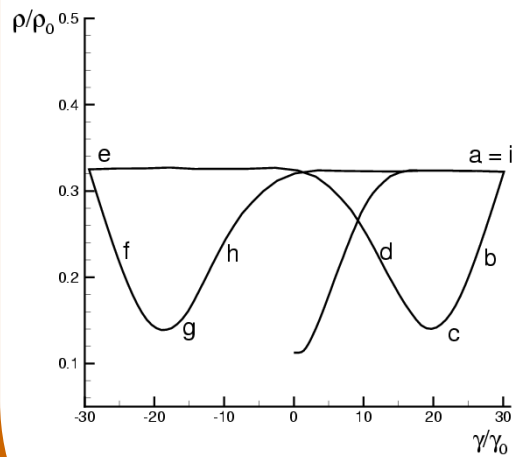
Stress-strain curve.



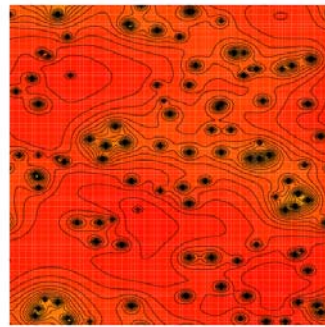
Cyclic loading



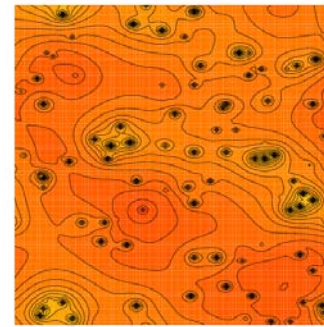
Stress-strain curve.



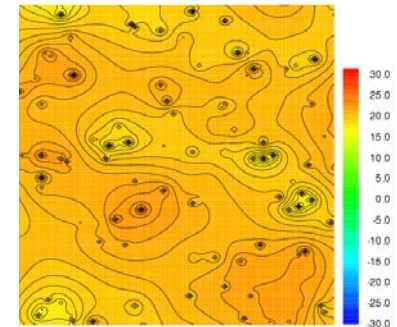
Evolution of dislocation density with strain.



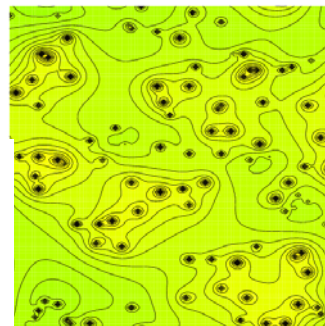
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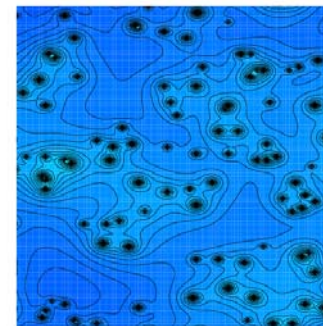
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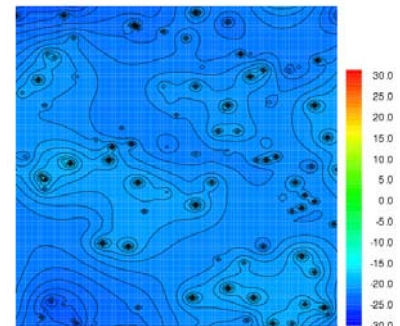
c



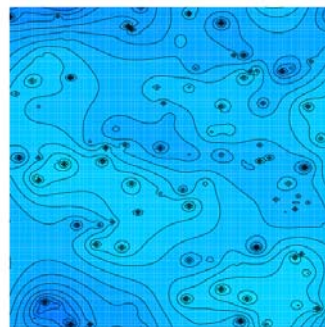
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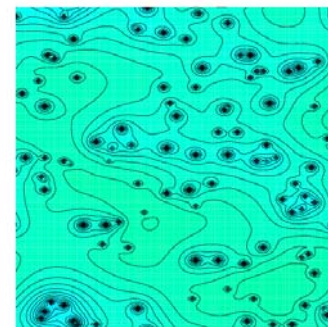
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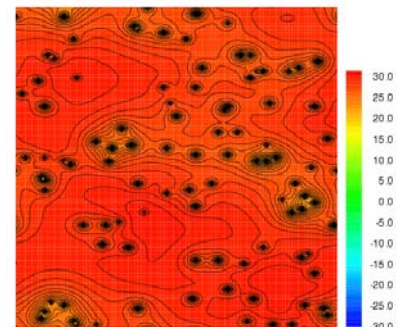
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g



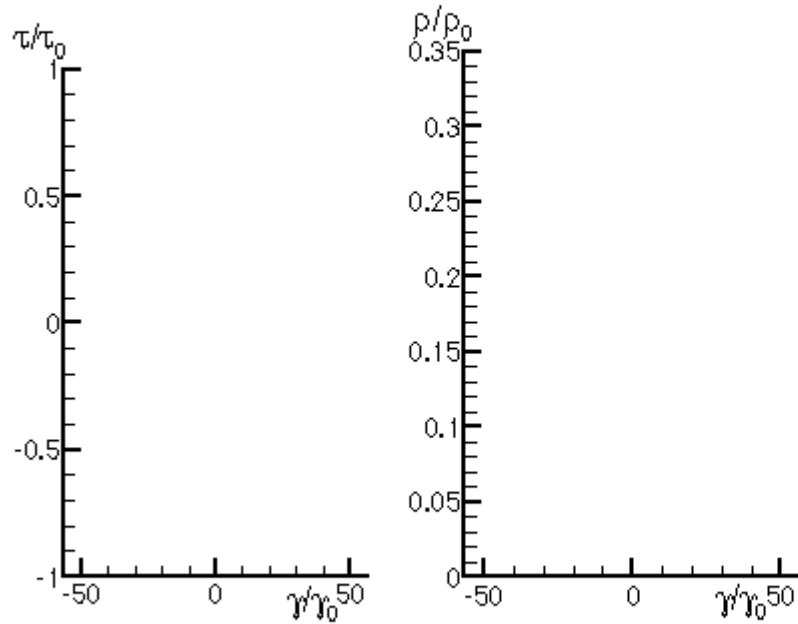
h



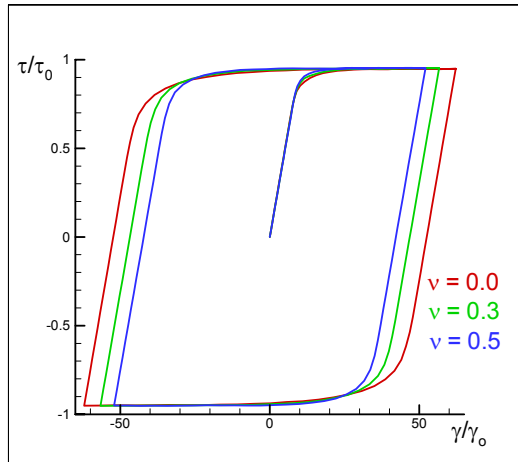
i



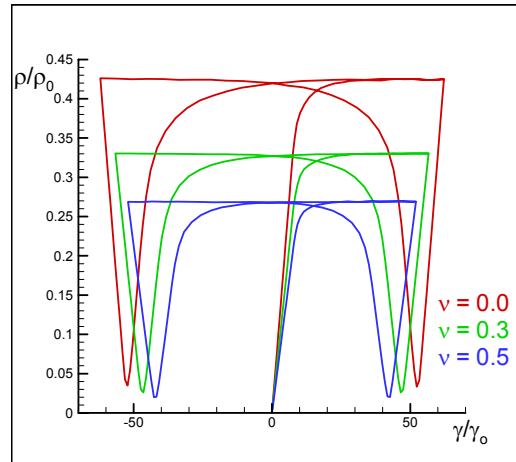
Cyclic loading



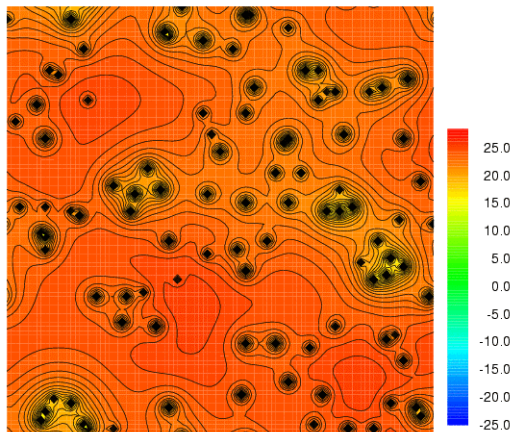
Poisson ratio effects



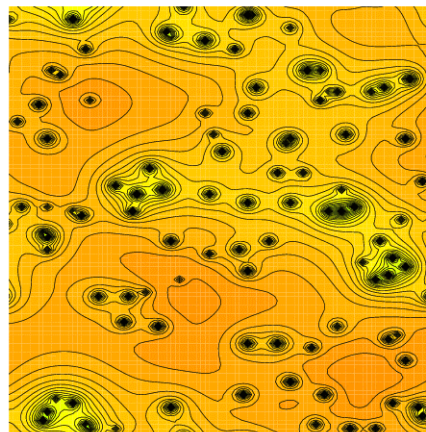
Stress-strain curve.



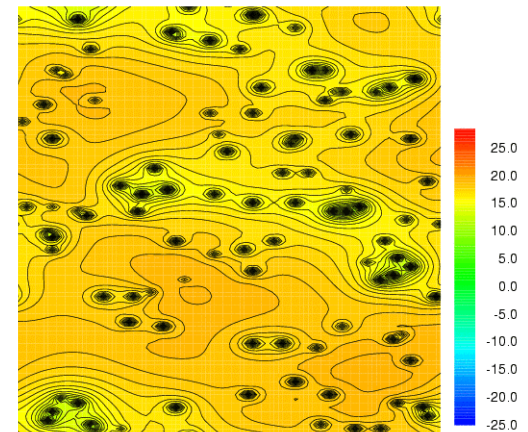
Evolution of dislocation density with strain.



$\nu = 0.0$



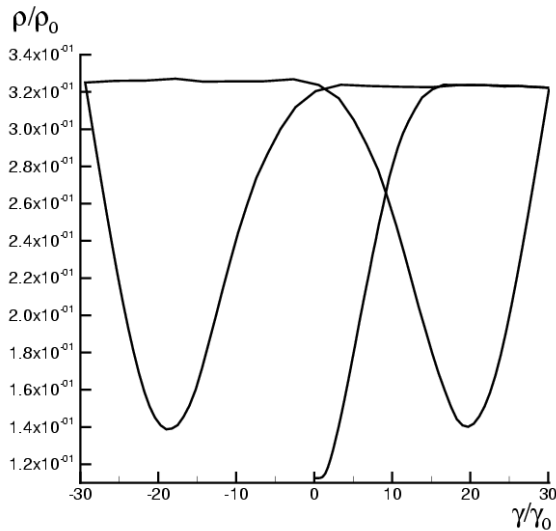
$\nu = 0.3$



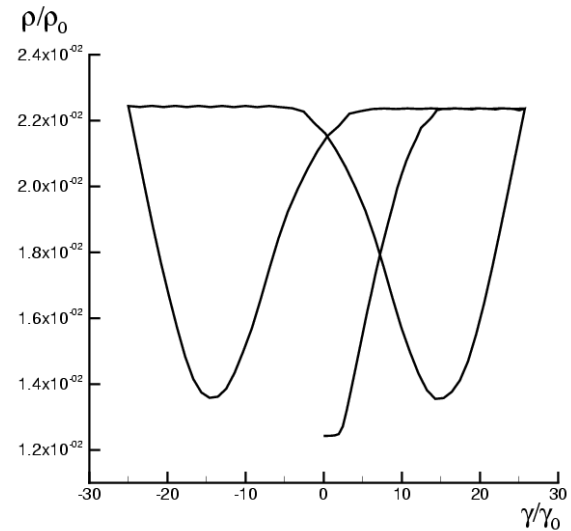
$\nu = 0.5$



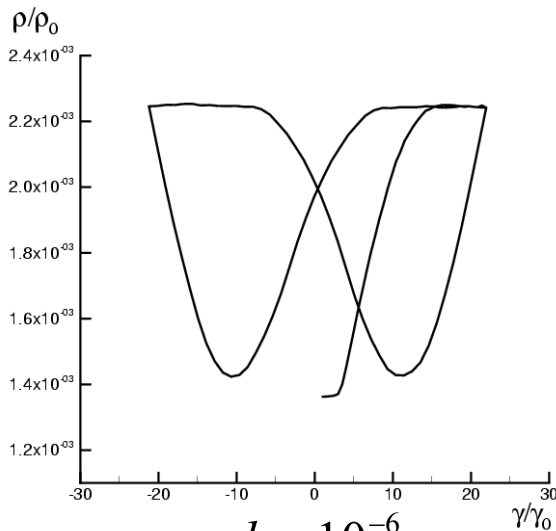
Obstacle density



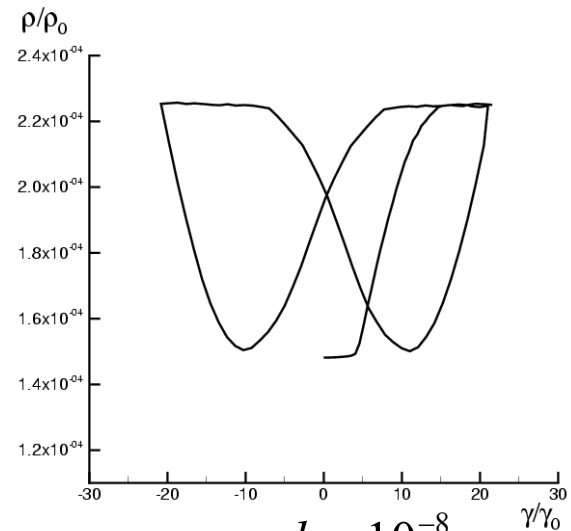
$$\rho b = 10^{-2}$$



$$\rho b = 10^{-4}$$



$$\rho b = 10^{-6}$$



$$\rho b = 10^{-8}$$



Temperature effects

- Incremental work function

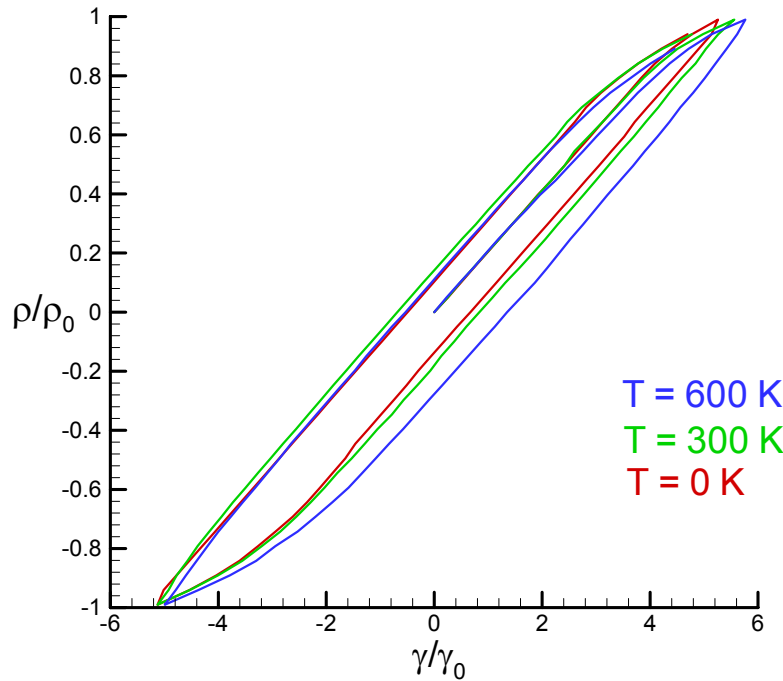
$$W[\xi^{n+1} | \xi^n] = E[\xi^{n+1}] - E[\xi^n] + \underbrace{\sum_{i=1}^{Nobs} F_i^{obs} |\xi^{n+1} - \xi^n|}_{\text{incremental work dissipated at the obstacles}}$$

- Metropolis Monte Carlo algorithm with hysteresis.
- Dislocation interaction is calculated in Fourier transformed space.

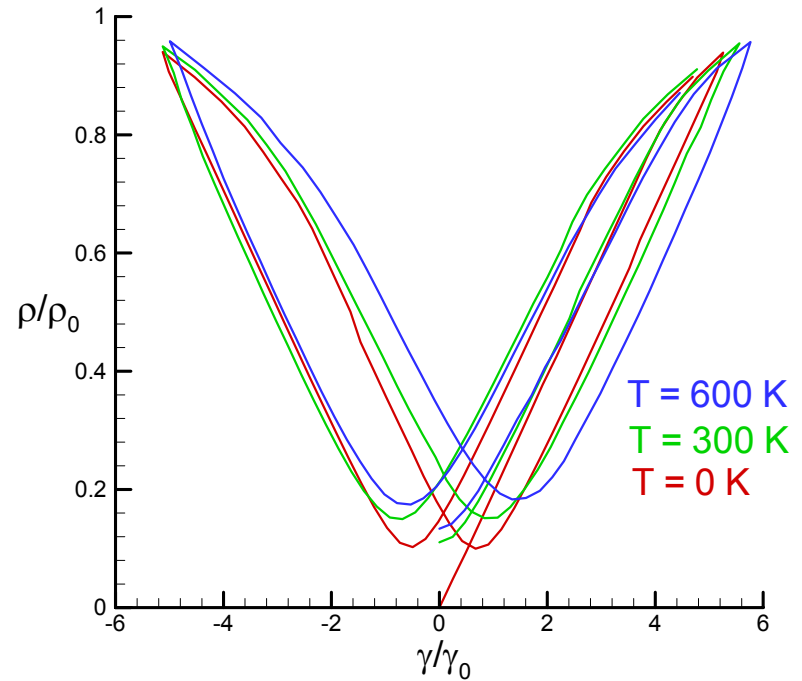
$$\mu = 5.6 \cdot 10^{10} \text{ Pa} \quad b = 2.5 \cdot 10^{-10} \text{ m}$$



Temperature Effects



Stress-strain curve.



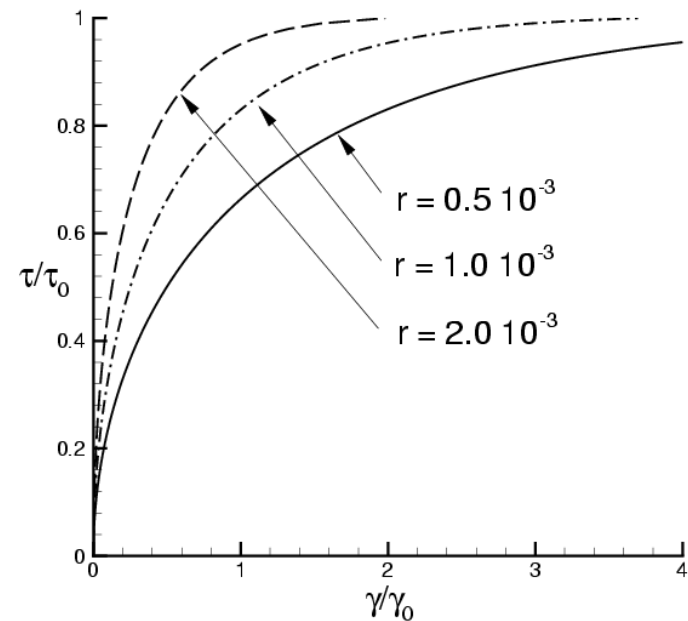
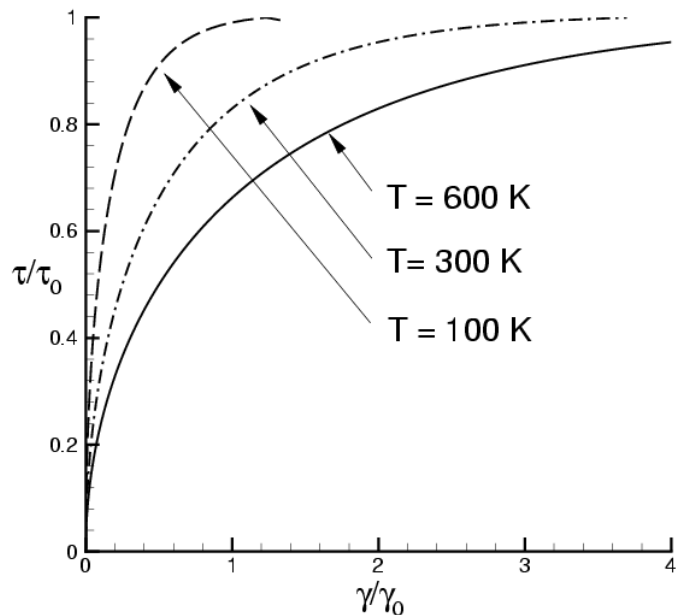
Evolution of dislocation density with strain.



Mean field approximation

$$\Delta w(\gamma_{n+1}, \gamma_n) = b^3 \left[-\tau_{n+1} \gamma_{n+1} + \tau_n \gamma_n + \tau_0 |\gamma_{n+1} - \gamma_n| \right]$$

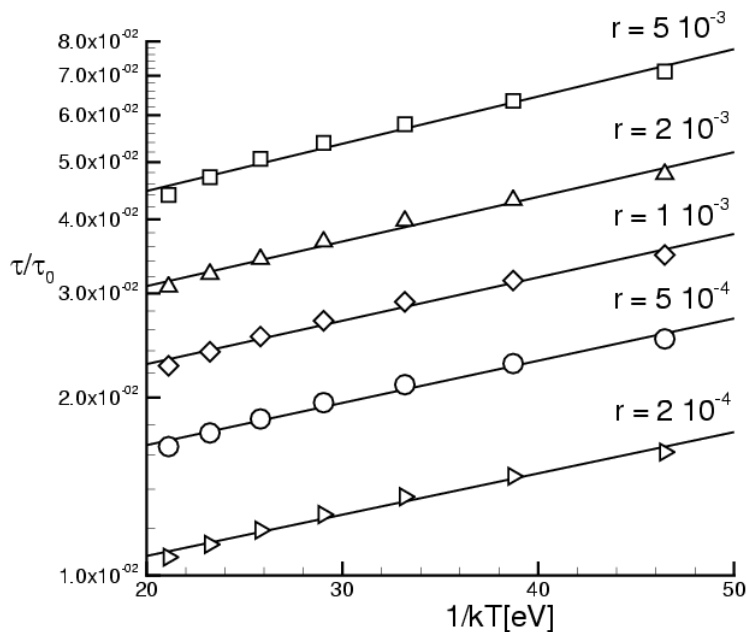
$$\langle \gamma_{n+1} \rangle = \langle \gamma_n \rangle + \frac{2\tau_{n+1}}{\beta b^3 (\tau_0^2 - \tau_{n+1}^2)}$$



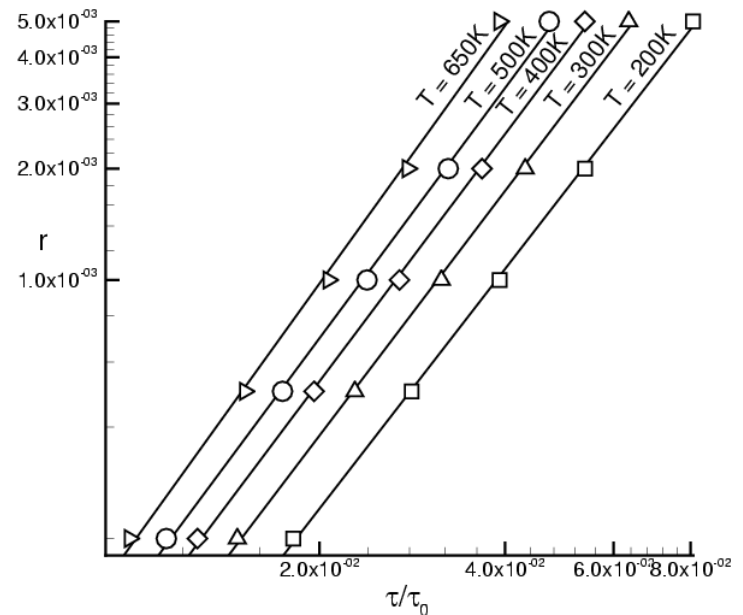
Mean field approximation

$$r = A \left(\frac{\tau}{\tau_0} \right)^n \exp \left(- \frac{U}{kT} \right)$$

(Kocks, Argon, Ashby, 1975)



Flow stress vs. temperature



Deformation rate vs. flow stress



Conclusions

- The phase-field representation furnishes a simple and effective means of tracking the motion of **large numbers of dislocations within discrete slip planes** through random arrays of obstacles (forest dislocations, defects) under the action of an applied shear stress.
- The theory predicts a range of behaviors which are in qualitative agreement with observation, **Bauschinger effect**, formation of **Orowan loops**.
- **Softening and yield stress** dependence on the temperature are observed.
- **Rate effects** are observed at finite temperature.

